



# Aqueous Synthesis of ZnO for Applications in Optoelectronics

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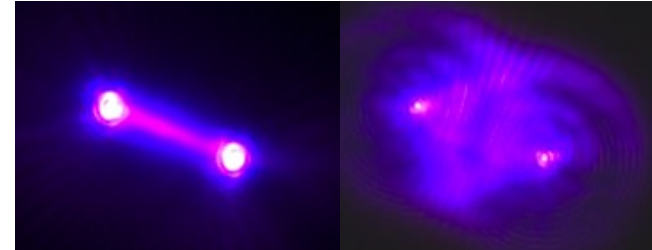
# ZnO for optoelectronics

Source

ZnO LED

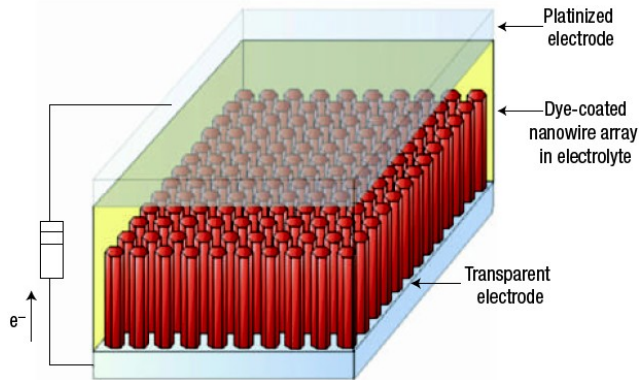


ZnO nanowire laser

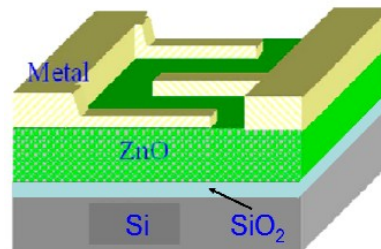


Detect

ZnO nanowire DSSC



ZnO photodetector



Manipulate

ZnO electro-optic modulator

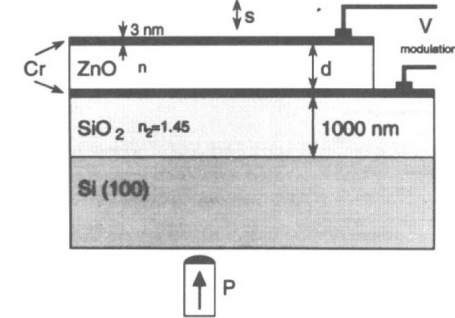


Image sources:

LED: <http://www.photonics.com/Article.aspx?AID=25012>

Laser: <http://versteegh.webnode.nl/research/>

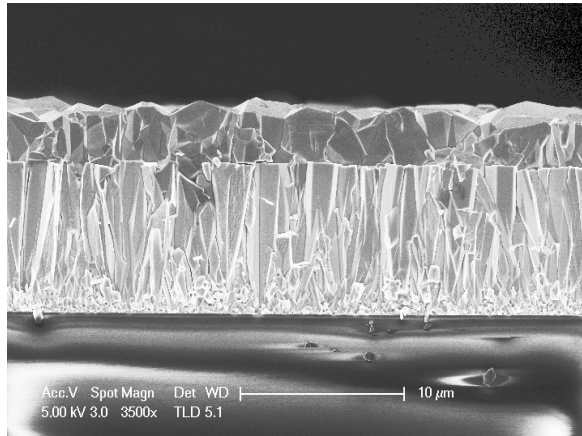
DSSC: Law, M.; Greene, L. E.; Johnson, J. C.; Saykally, R.; Yang, P. Nat Mater 2005, 4, 455–459.

Modulator: Koch, M. H.; Timbrell, P. Y.; Lamb, R. N. Semicond. Sci. Technol. 1995, 10, 1523.

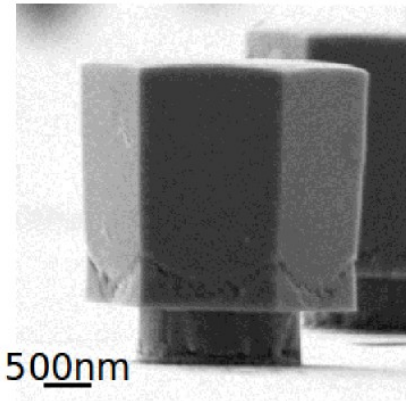
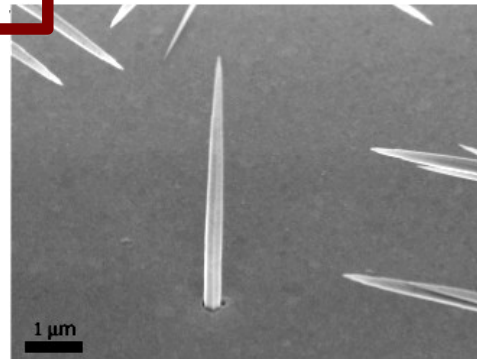
Photodetector: Liu, K.; Sakurai, M.; Aono, M. Sensors 2010, 10, 8604–8634.



# Expanding the use of ZnO in optoelectronics

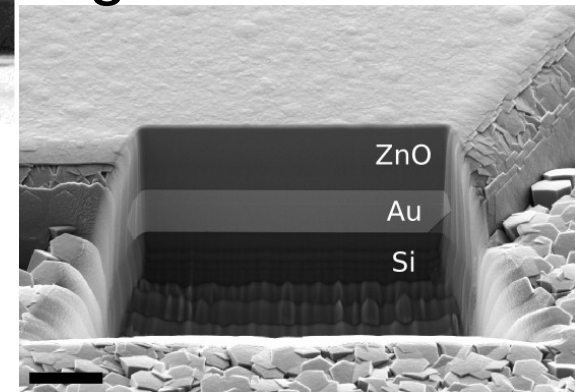


ZnO/Cu<sub>2</sub>O  
solar cells



Templated growth

Epitaxial  
growth on Au





Motivation

Device fabrication

Measurement techniques

Results

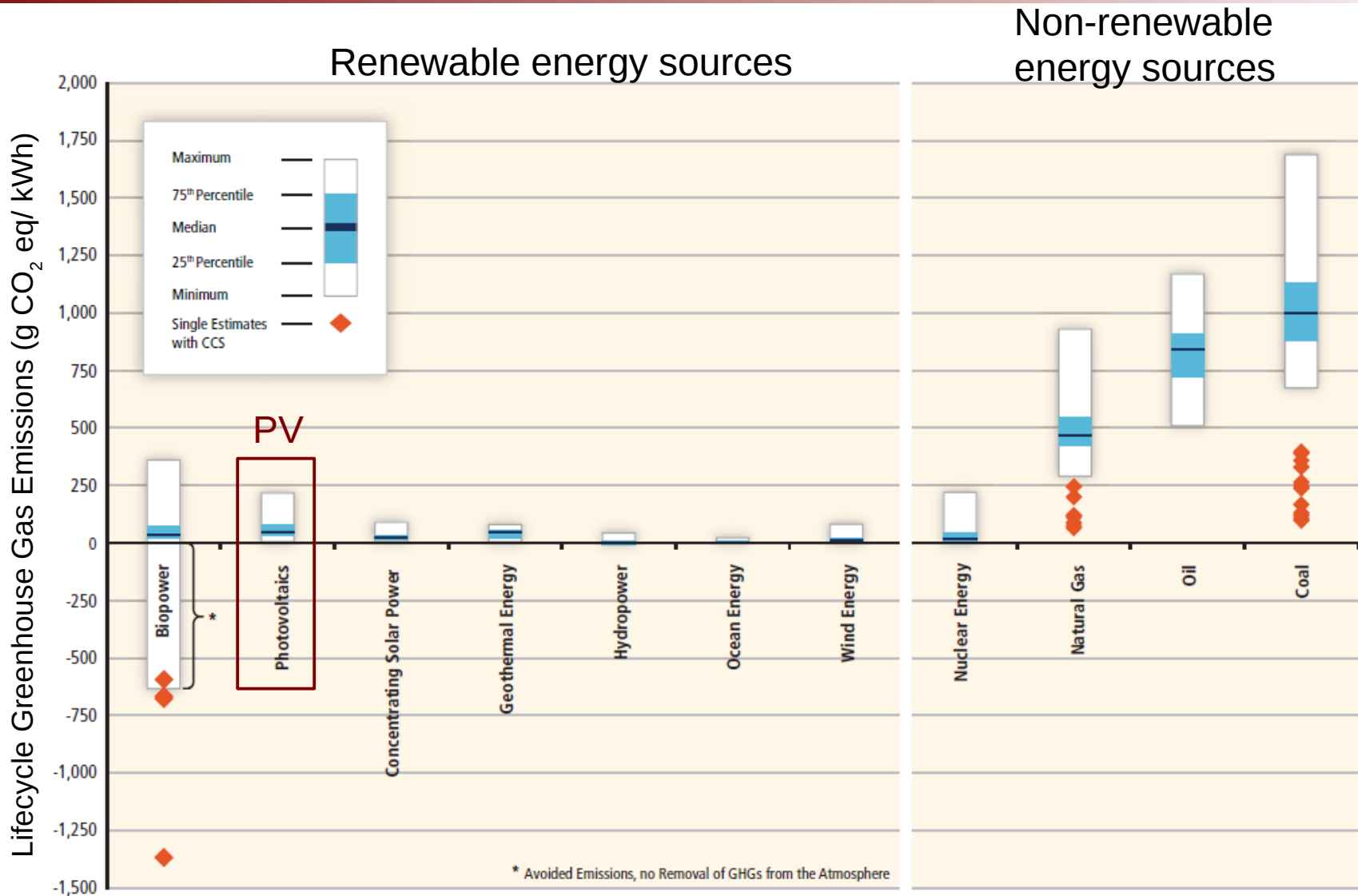
- Increased carrier collection
- Changing behavior over time

Discussion of mobile dopants

Summary and Implications

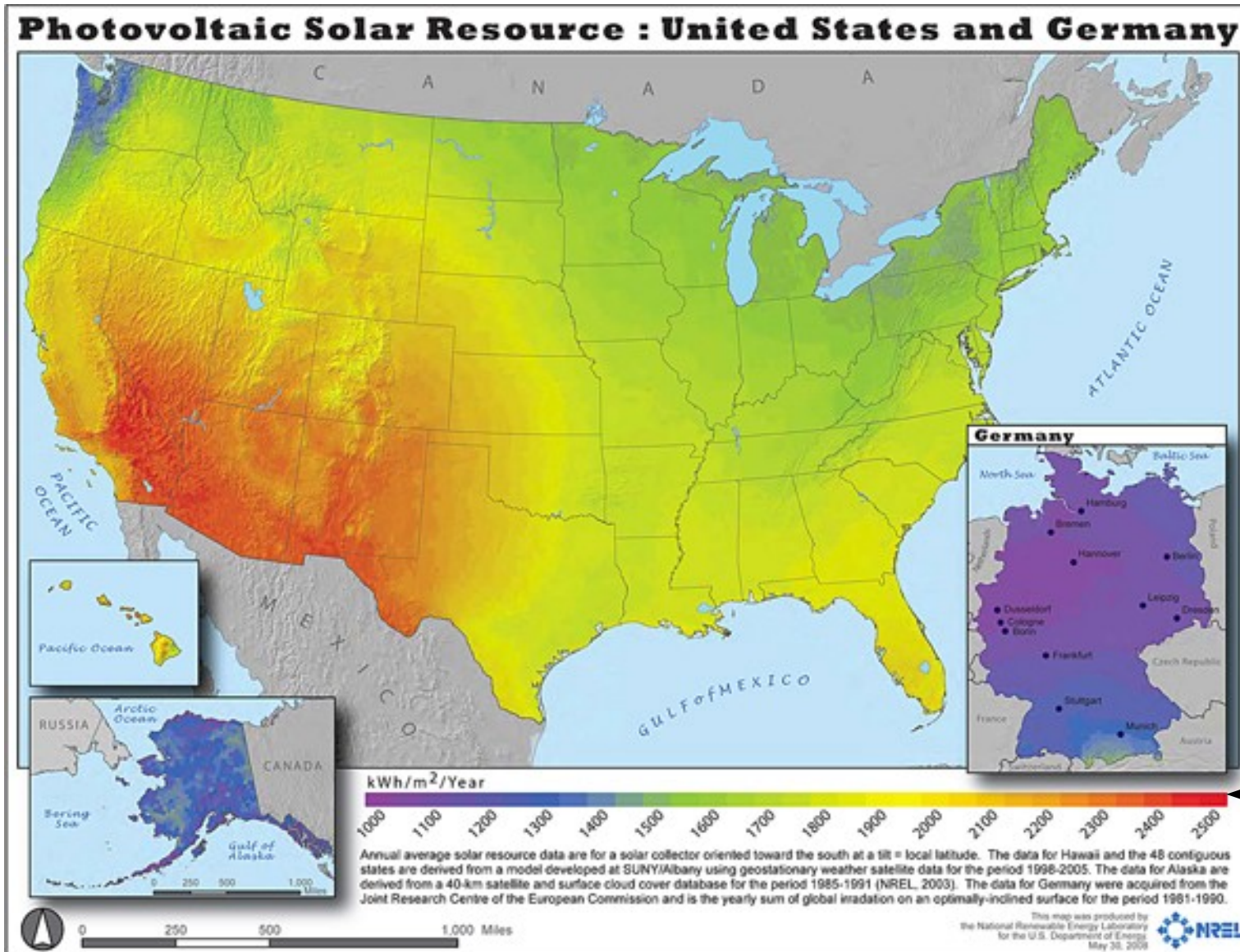


# Benefits of using solar cells





# Benefits of using solar cells



kWh/m²/Year



# Benefits of using solar cells

**Table 3.1** | Annual total technical potential of solar energy for various regions of the world, not differentiated by conversion technology (Rogner et al., 2000; their Table 5.19).

REGIONS	Range of Estimates	
	Minimum, EJ	Maximum, EJ
North America	181	7,410
Latin America and Caribbean	113	3,385
Western Europe	25	914
Central and Eastern Europe	4	154
Former Soviet Union	199	8,655
Middle East and North Africa	412	11,060
Sub-Saharan Africa	372	9,528
Pacific Asia	41	994
South Asia	39	1,339
Centrally planned Asia	116	4,135
Pacific OECD	73	2,263
<b>TOTAL</b>	<b>1,575</b>	<b>49,837</b>
Potential global solar energy/Global energy demand →	<b>3.2</b>	<b>101</b>

Note: Basic assumptions used in assessing minimum and maximum technical potentials of solar energy are given in Rogner et al. (2000):

- Annual minimum clear-sky irradiance relates to horizontal collector plane, and annual maximum clear-sky irradiance relates to two-axis-tracking collector plane; see Table 2.2 in WEC (1994).
- Maximum and minimum annual sky clearance assumed for the relevant latitudes; see Table 2.2 in WEC (1994).



# Benefits of using solar cells

- Clean energy source
- Domestic energy source
- Abundant energy source
- Save petroleum for other of humanity's needs

Solvents

Bearing Grease

Ink

Floor Wax

Ballpoint Pens

Football Cleats

Upholstery

Sweaters

Boats

Insecticides

Bicycle Tires

Sports Car Bodies

Nail Polish

Fishing lures

Dresses

Tires

Golf Bags

Perfumes

Cassettes

Purses

Dishwasher parts

Tool Boxes

Shoe Polish

Motorcycle Helmet

Caulking

Petroleum Jelly

Transparent Tape

CD Player

Faucet Washers

Antiseptics

Clothesline

Curtains

Food Preservatives

Basketballs

Soap

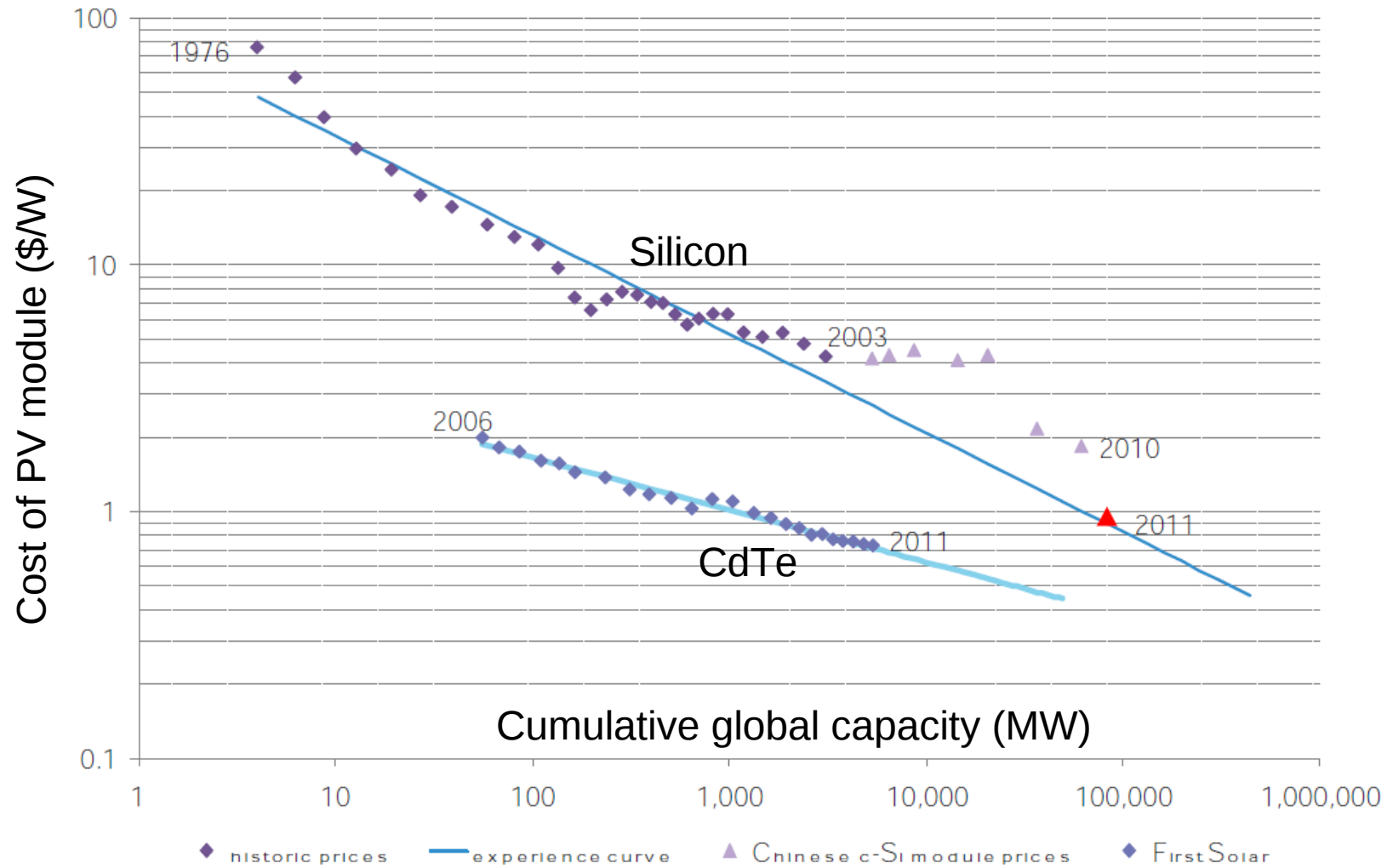
Vitamin Capsules

Antihistamines

Aspirin

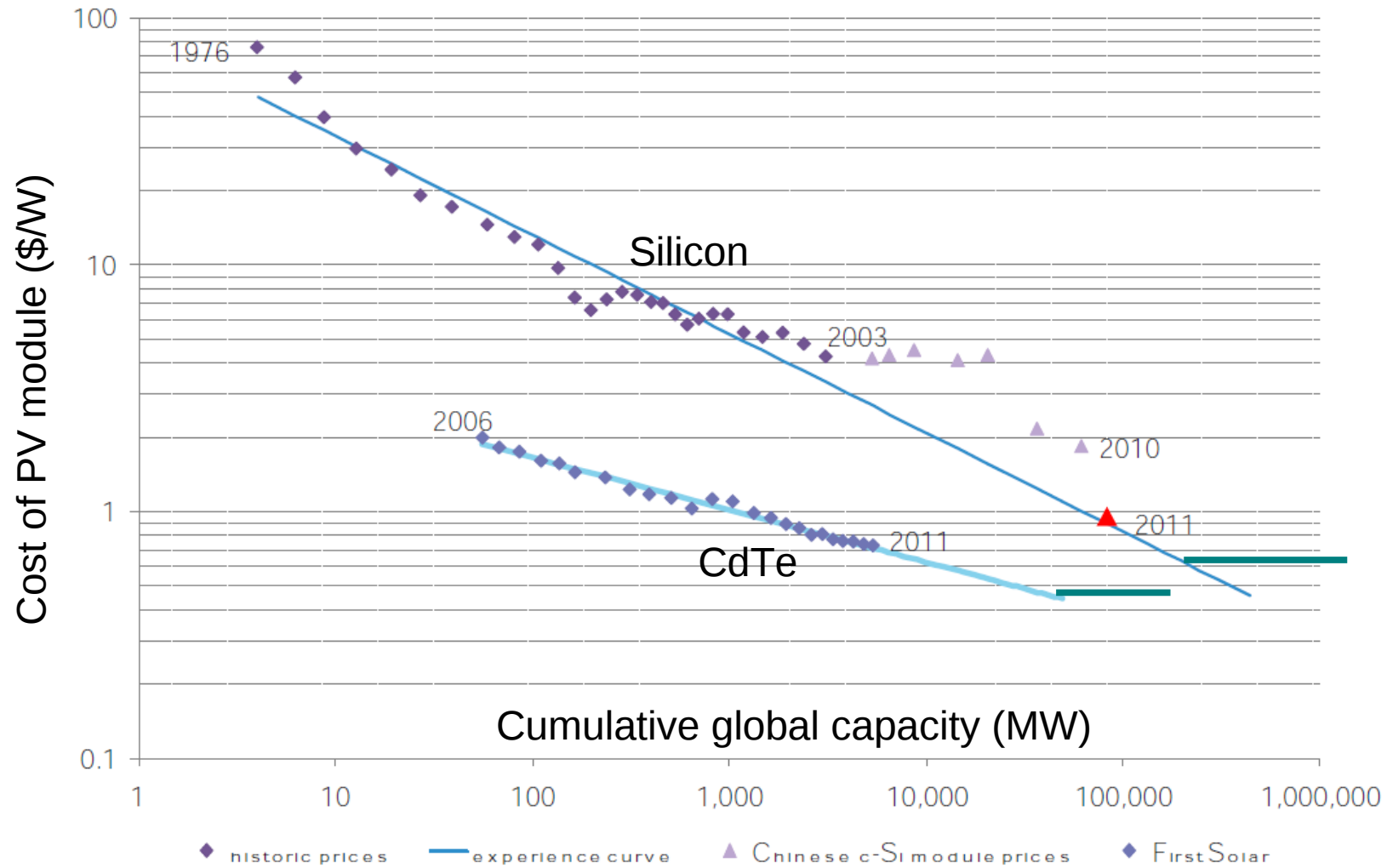


# Current economic state of solar cells





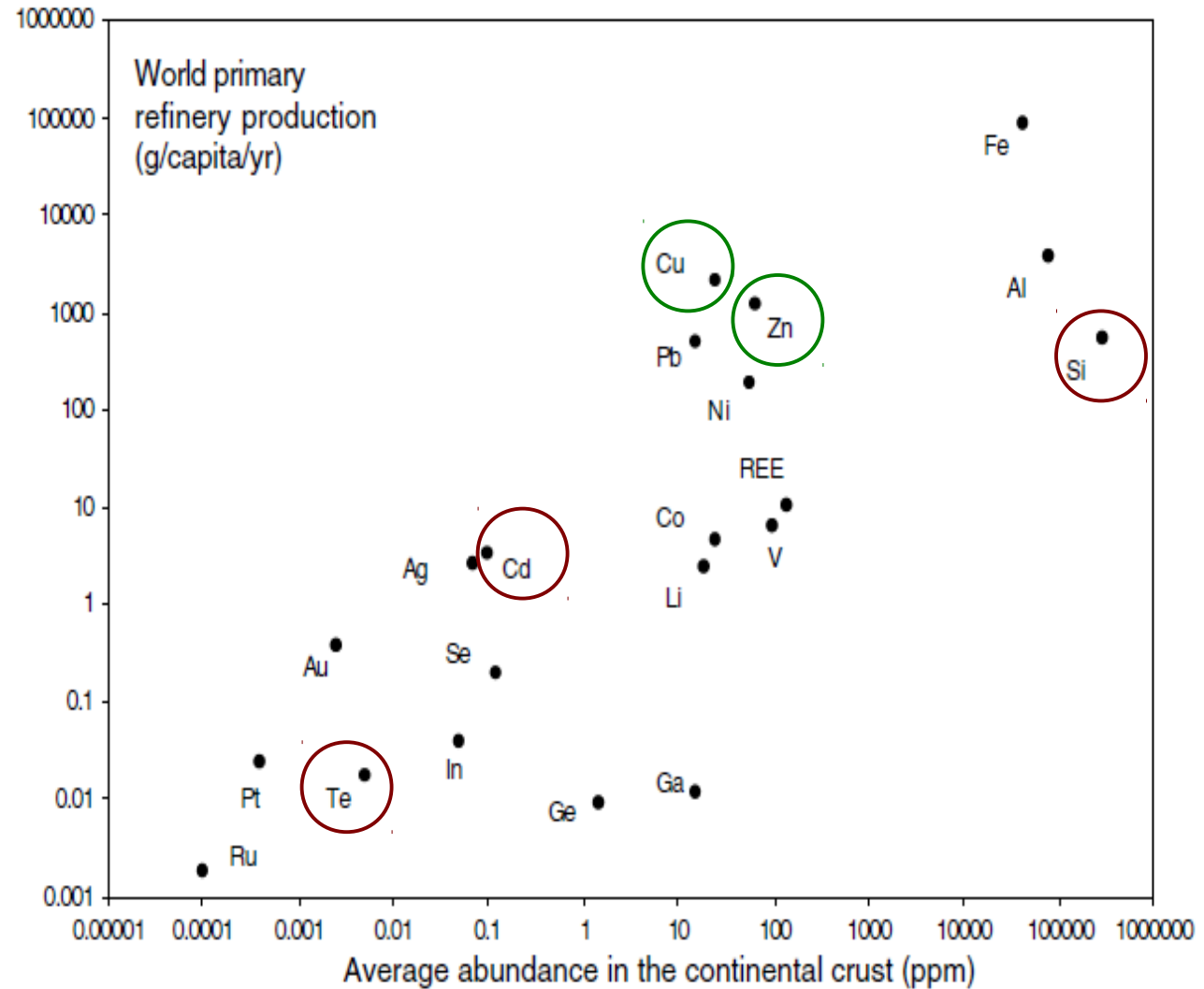
# Current economic state of solar cells





# Limitations in materials

Predominant ore	$\Delta G_f$ (kJ/mol)
$\text{SiO}_2$	-856.4
$\text{CuFeS}_2$	-224.5





## Cu<sub>2</sub>O

- $E_g = 2.0$  eV
- Self-doped p-type material by  $V_{cu}$
- Inexpensive, abundant
- Easy to deposit polycrystalline through electrodeposition



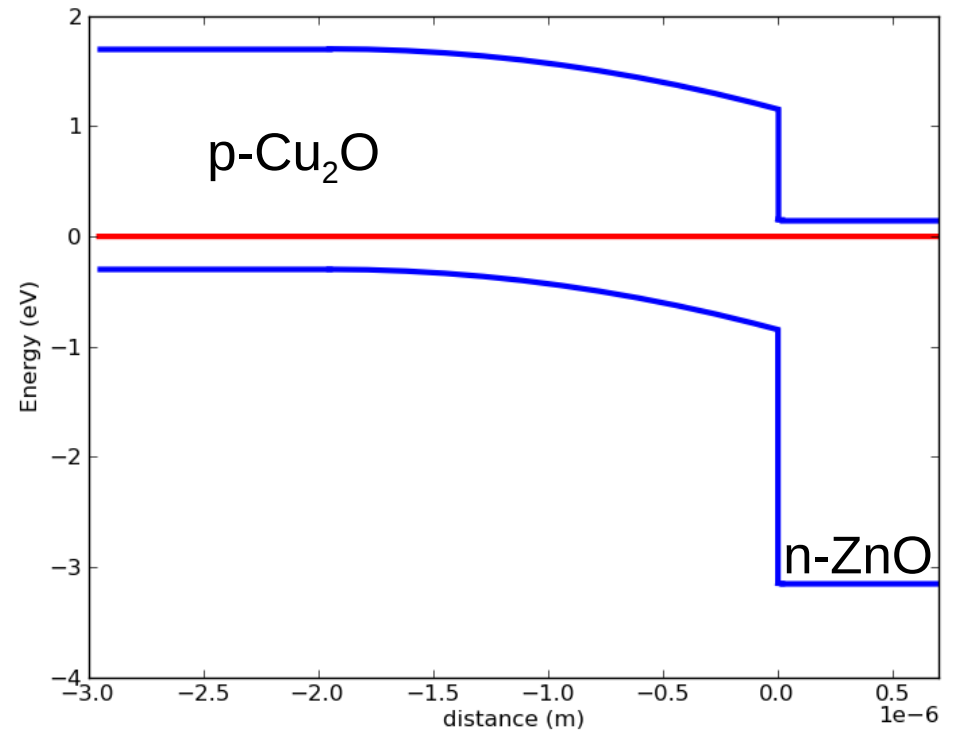
# Cu<sub>2</sub>O/ZnO

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## ZnO

- n-type doping via hydrogen
- Wide bandgap  $E_g = 3.4$  eV
- Type II heterojunction





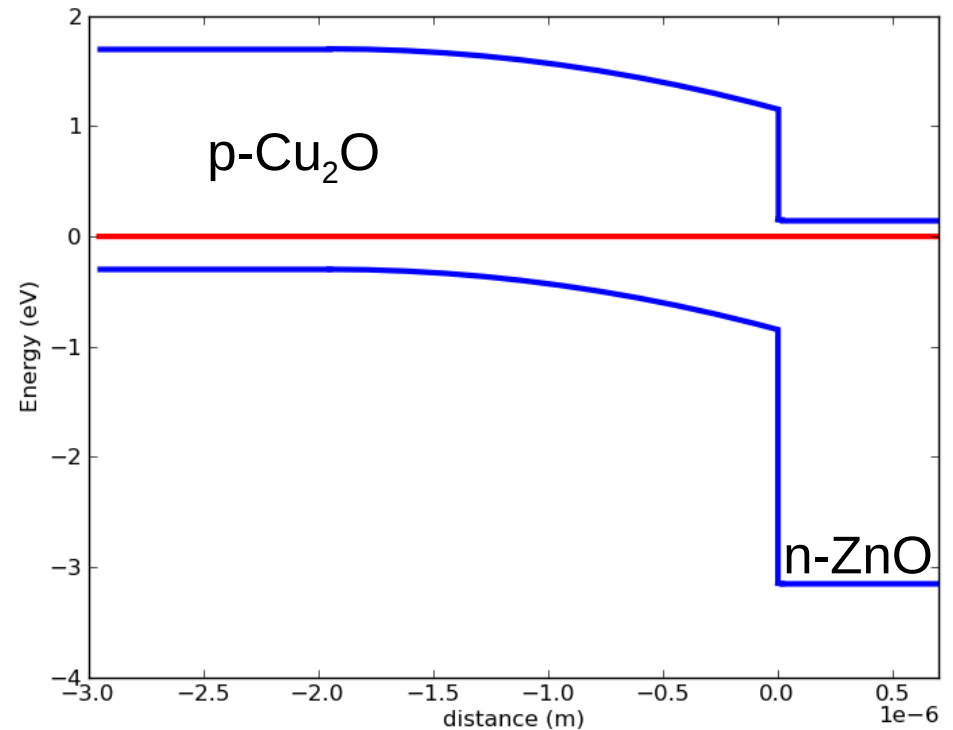
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## ZnO

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*Theoretical efficiency ~13%.  
Current record ~4%. [1]*



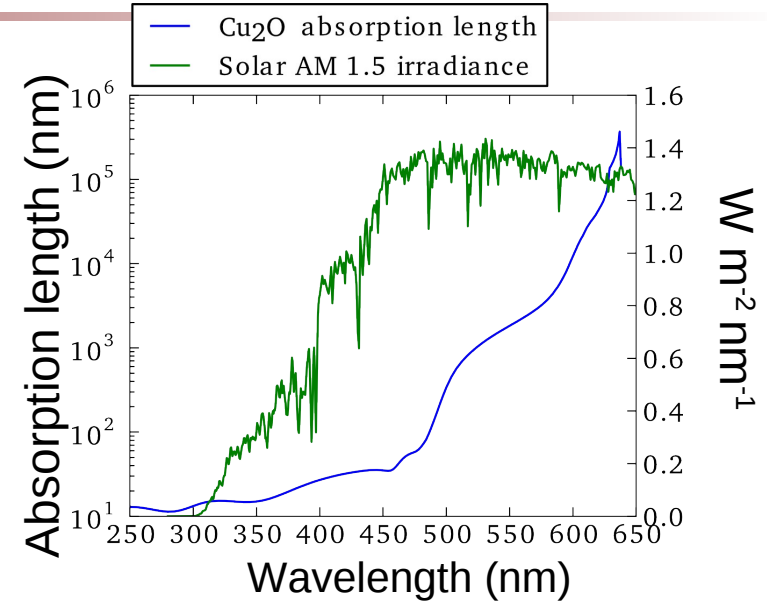
# Radial junction/3D solar cells

Absorption length =  $\alpha_L$

Minority carrier diffusion length =  $L_{min}$

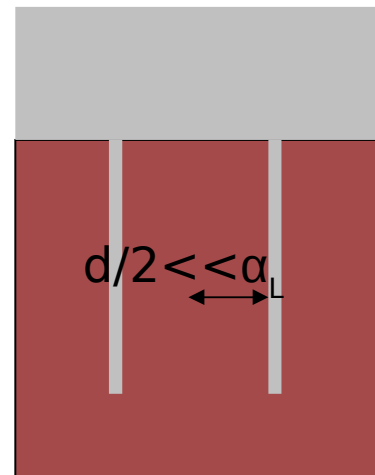
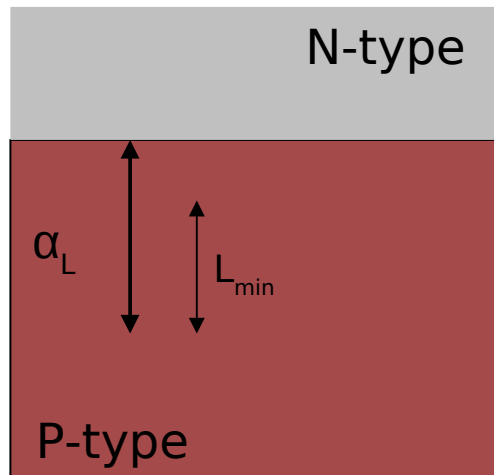
Recombination → Lost photocurrent

A common design constraint: Total thickness <  $L_{min}$



For electrodeposited Cu<sub>2</sub>O:

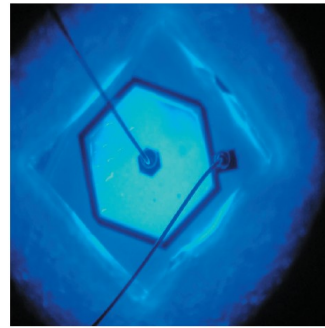
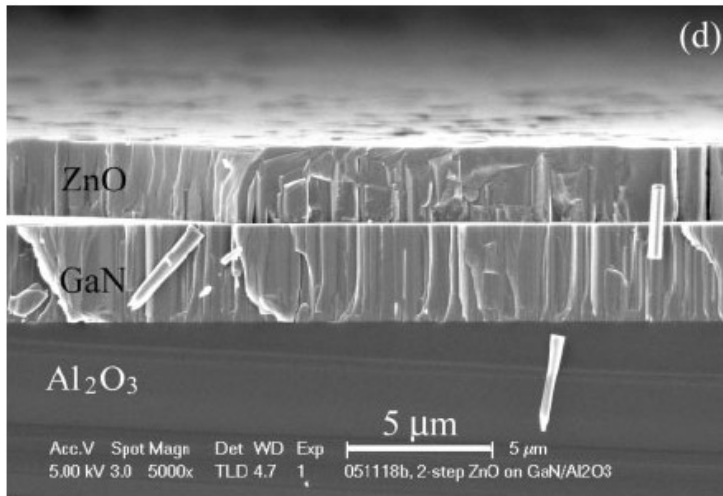
$L_{min} \sim 100\text{nm}$  [1]



(1) Musselman, K. P.; Ilevskaya, Y.; MacManus-Driscoll, J. L. Applied Physics Letters 2012, 101, 253503–253503–5.  
 (2) Absorption length derived from data in Malerba, C.; Biccari, F.; Leonor Azanza Ricardo, C.; D’Incau, M.; Scardi, P.; Mittiga, A. Solar Energy Materials and Solar Cells 2011, 95, 2848–2854.

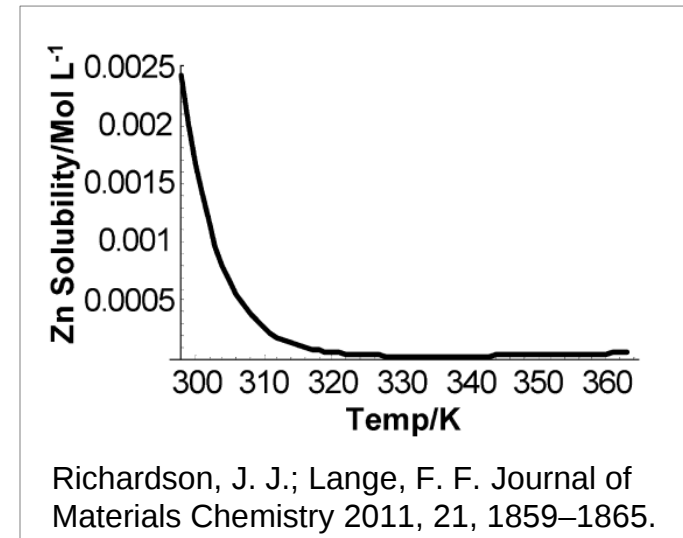
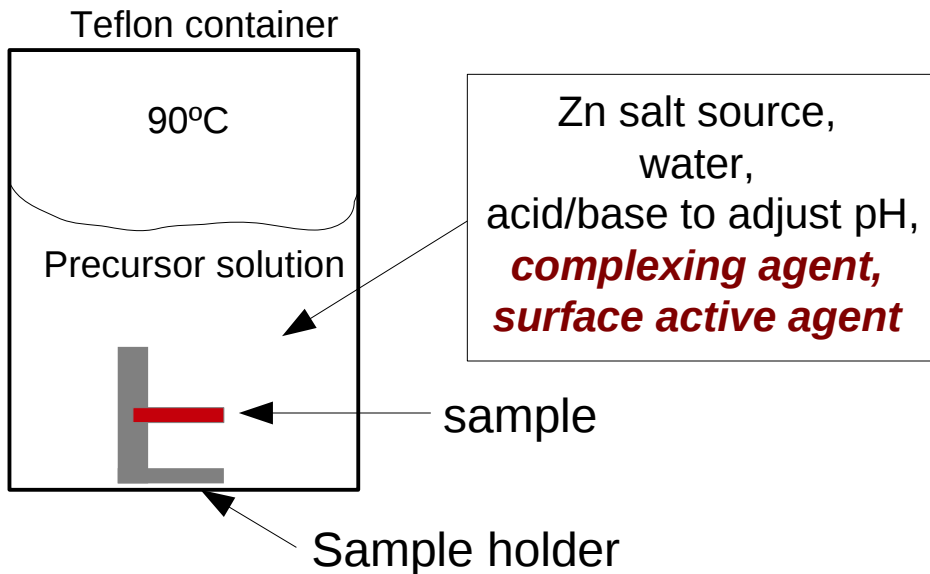
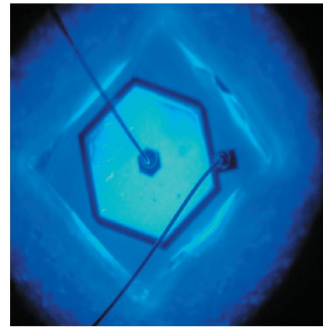
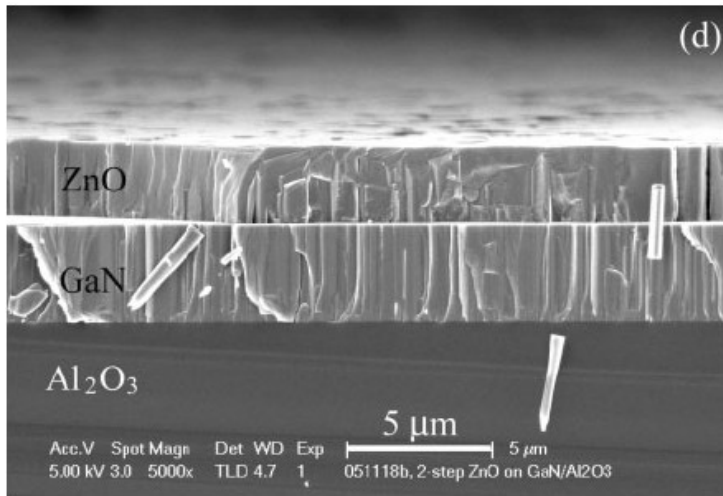


# ZnO aqueous solution growth





# ZnO aqueous solution growth

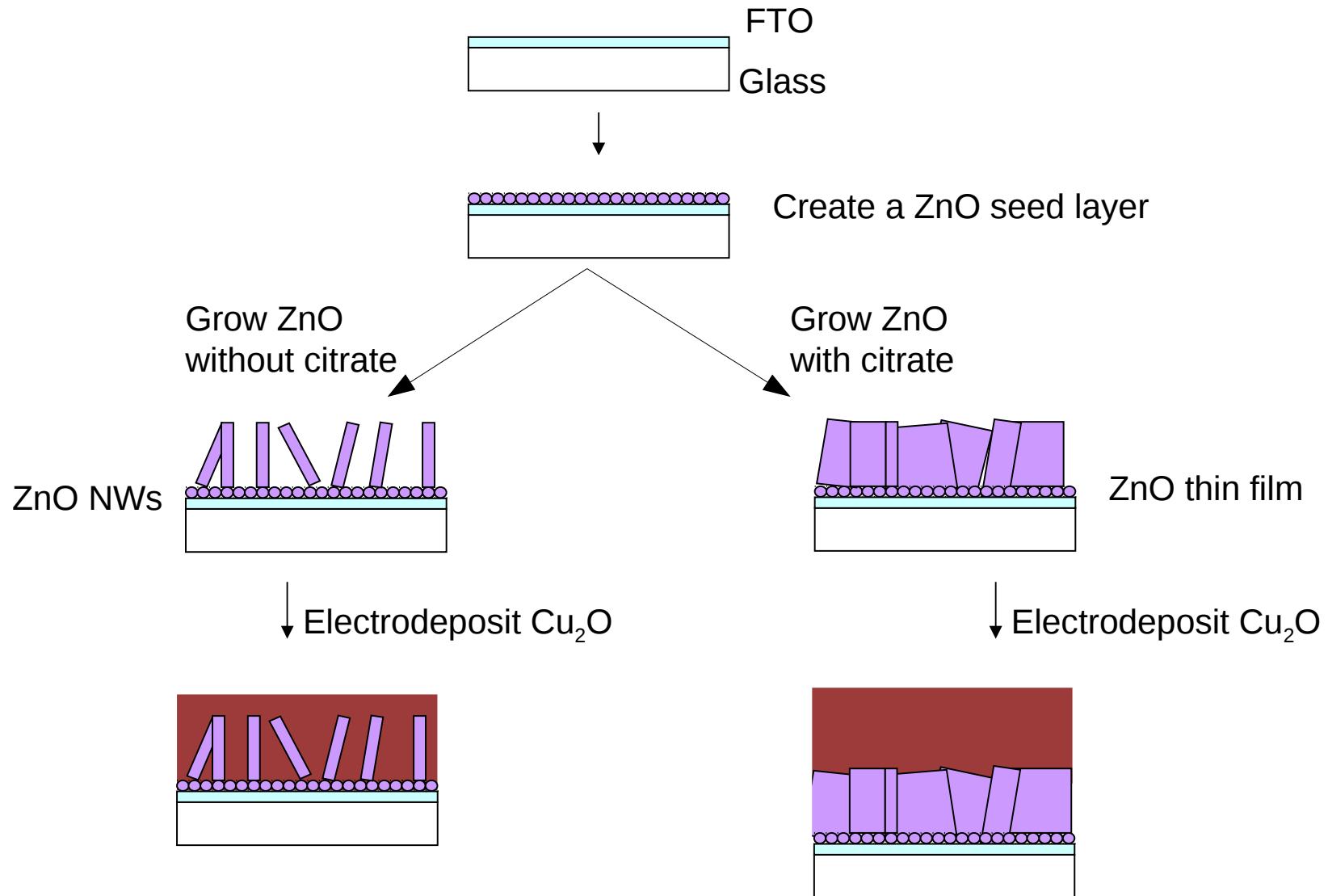


With addition of **ammonia**, zinc solubility decreases with increasing temperature!

Richardson, J. J.; Lange, F. F. Crystal Growth & Design 2009, 9, 2570–2575.

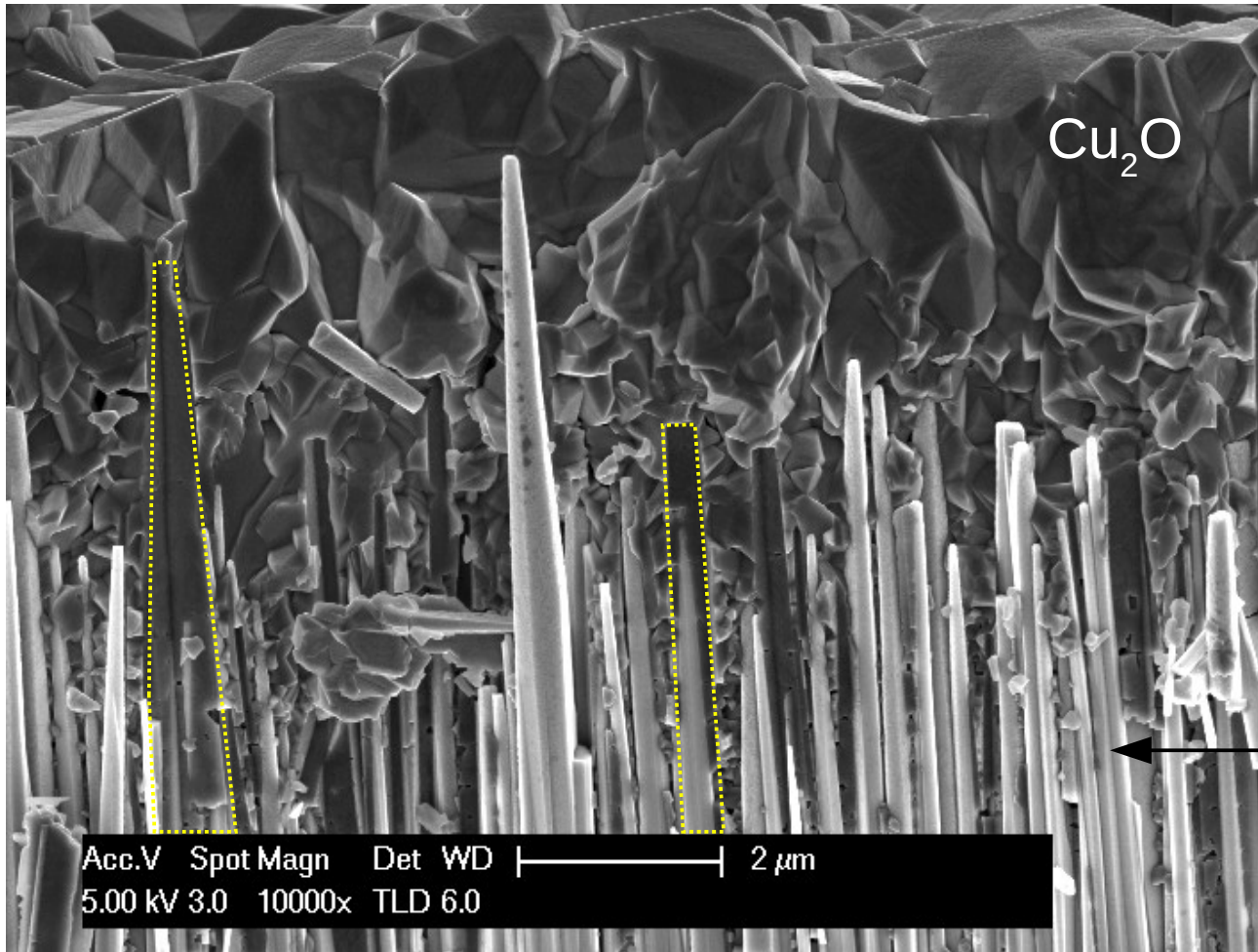


# Plan of making ZnO/Cu<sub>2</sub>O solar cells





# First attempts at ZnO/Cu<sub>2</sub>O solar cells

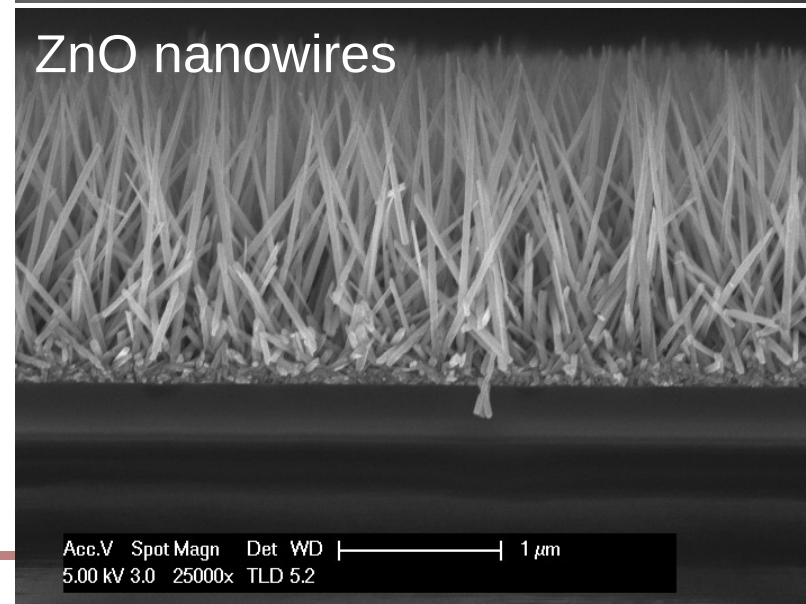
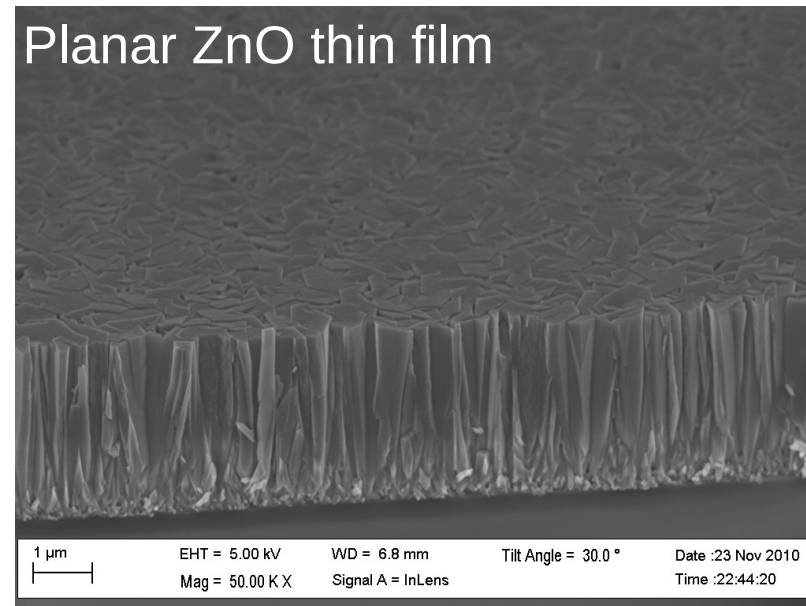
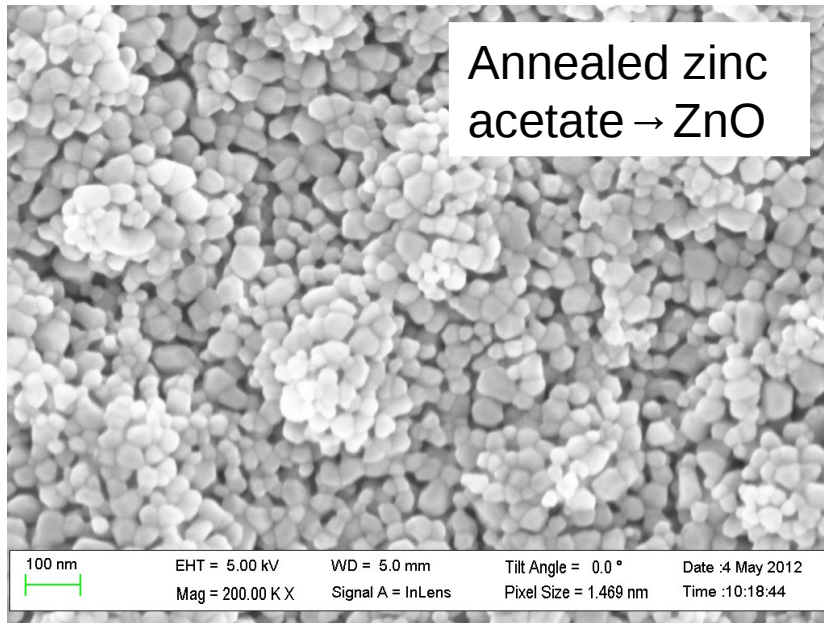


Not enough  
space between  
ZnO wires for  
Cu<sub>2</sub>O.

← ZnO wires



# Changed seeding to make nanowires

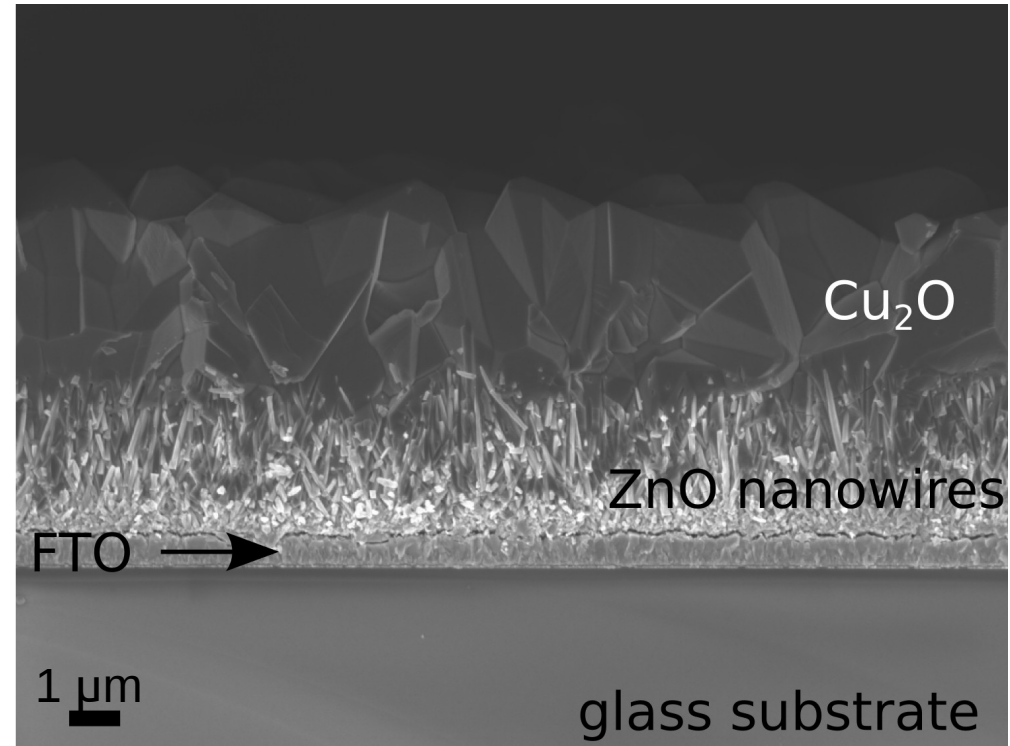
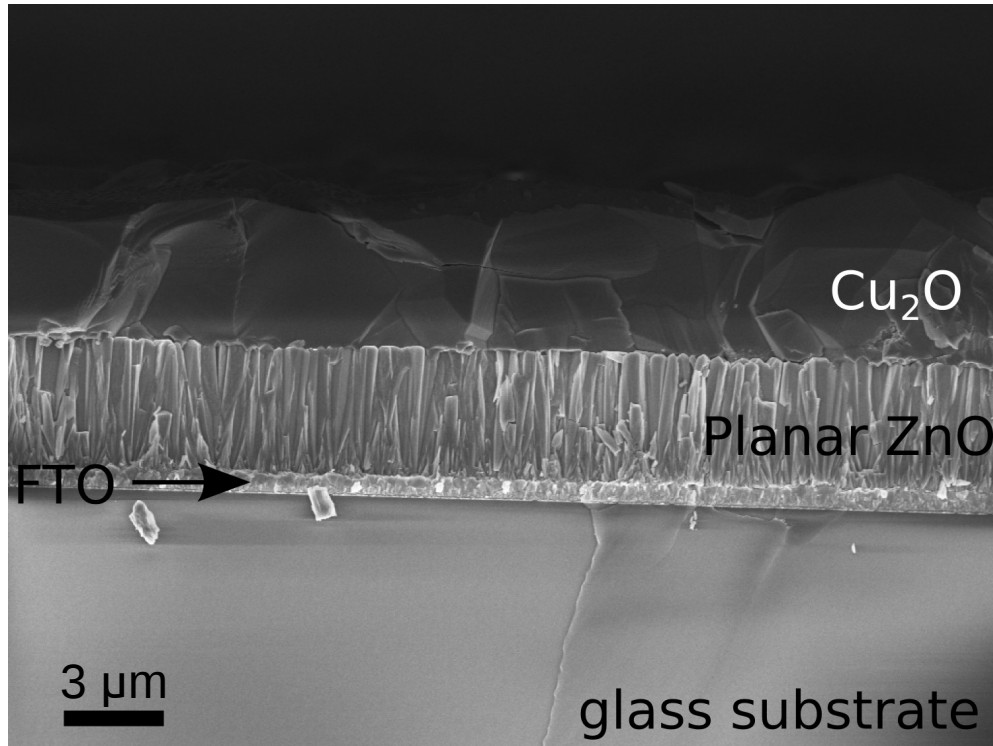


Seed size controls the diameter of the nanowire.

*Thinner wires!*

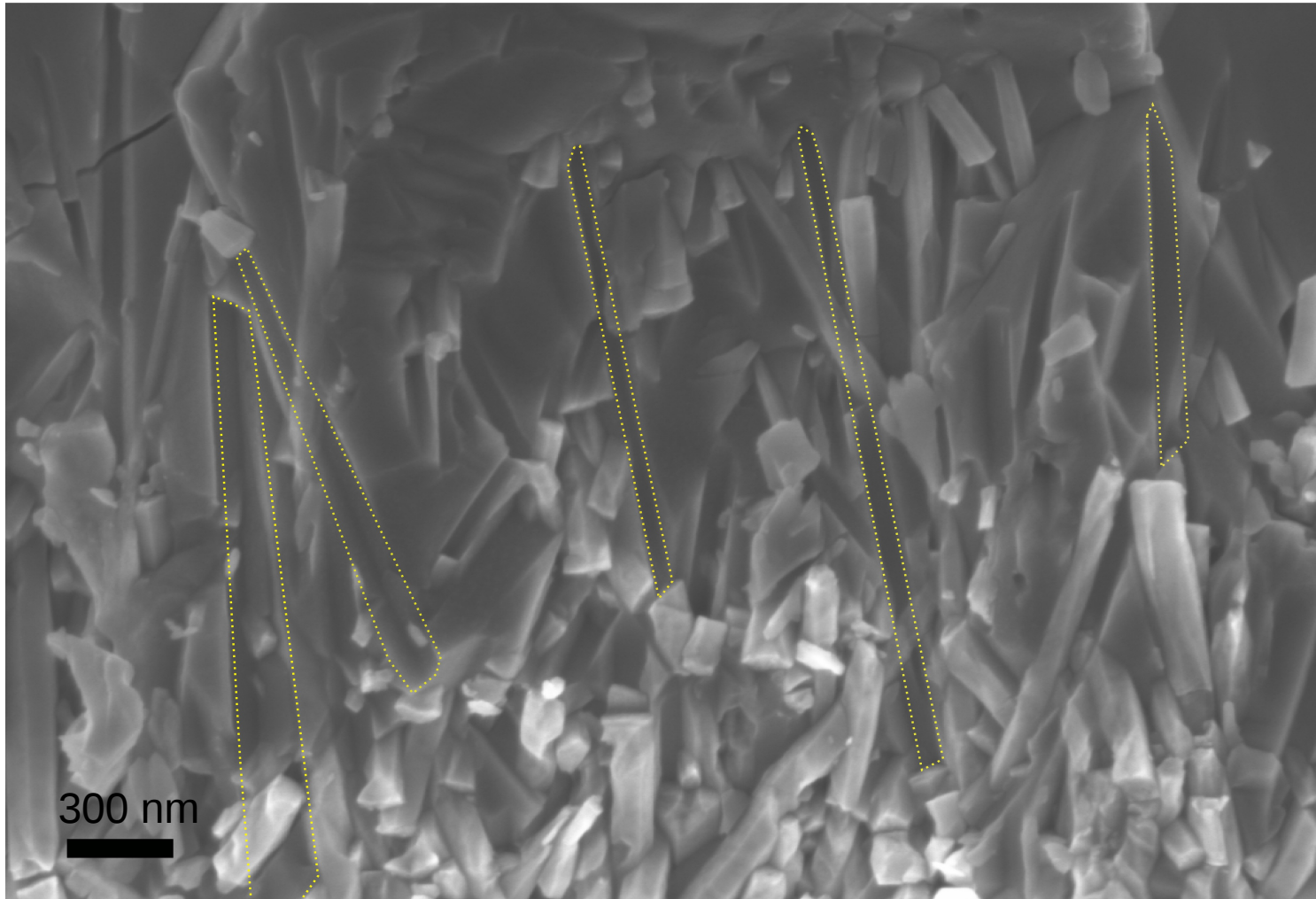


# Planar and NW solar cells





# NW impressions in $\text{Cu}_2\text{O}$ matrix





# Ideal solar cell formation

Form a p-n junction

n-type

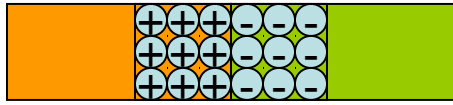
p-type



# Internal electric field in a p-n junction

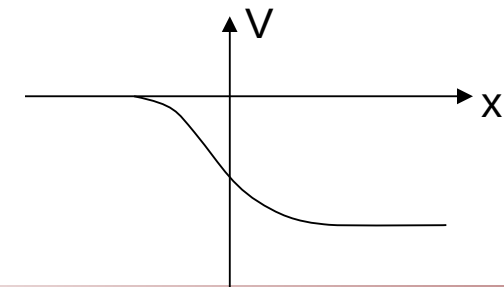
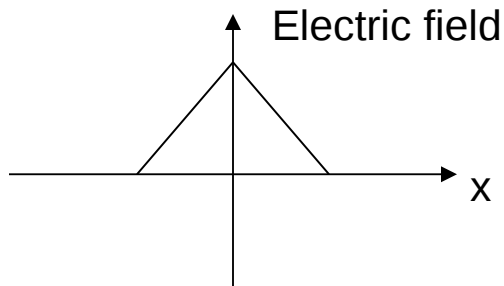
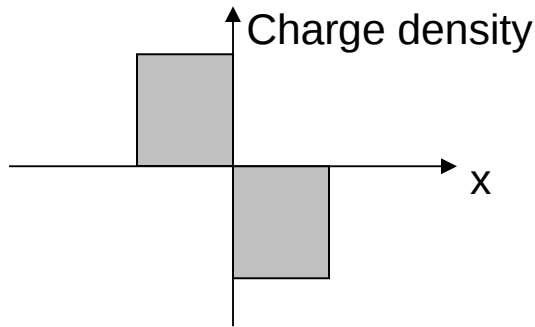
## Form a p-n junction

N-type  
(ZnO)



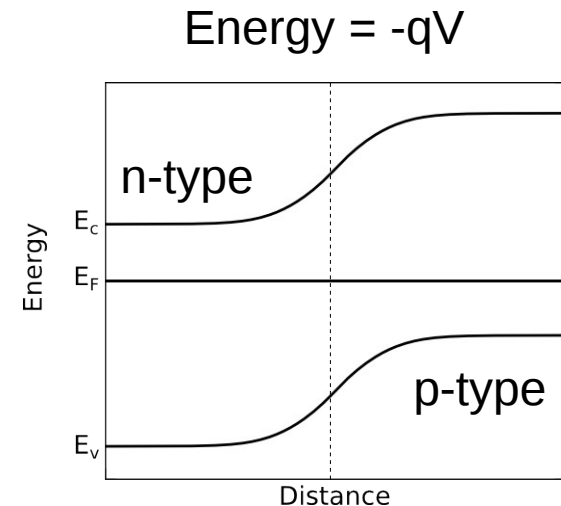
P-type  
(Cu<sub>2</sub>O)

As electrons and holes move to achieve equilibrium, they leave behind charged ions.



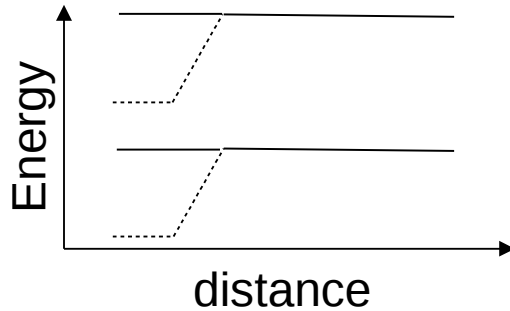
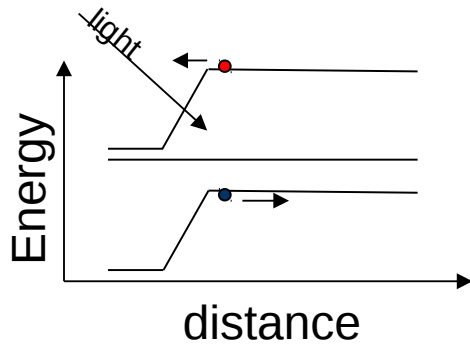
$$\frac{d\vec{E}}{dx} = \frac{\rho}{\epsilon}$$

$$\frac{dV}{dx} = -\vec{E}$$



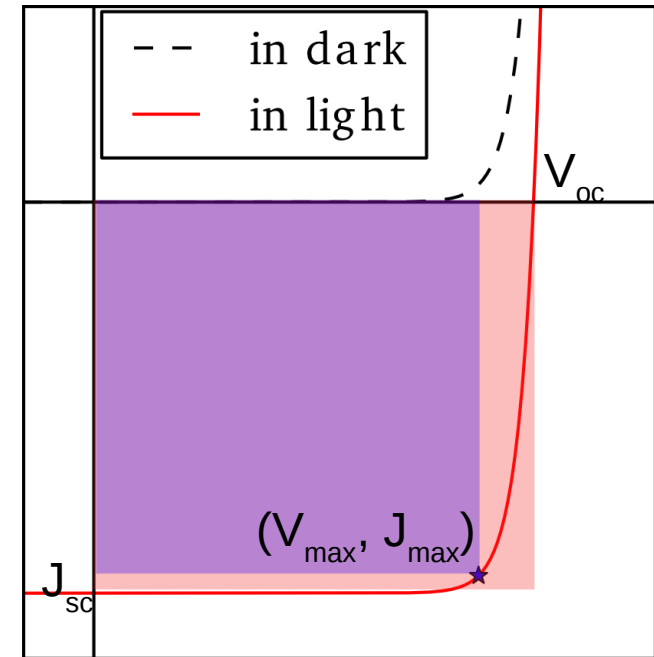


# Measuring current-voltage



- $I_{sc}$  : short circuit current.
- Measure of photon absorption and carrier collection.
- $V_{oc}$  : open circuit voltage.
- Measure of built-in voltage.

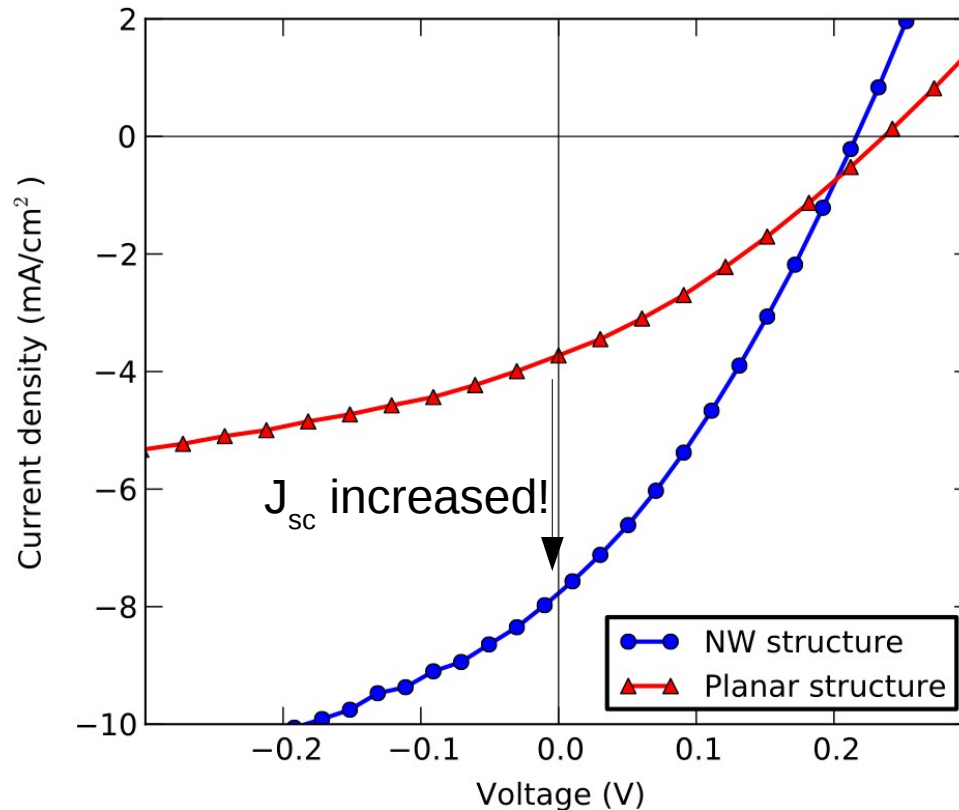
Fill factor,  $FF = (J_{max} V_{max}) / (J_{sc} V_{oc})$



$$P_{max} = J_{sc} V_{oc} FF$$



# Enhanced carrier collection: $J_{sc}$



## Planar ZnO on FTO

$J_{sc} = 4.1 \text{ mA/cm}^2$

$V_{oc} = 0.236 \text{ V}$

FF = 0.305

Eff = 0.30%

## NW ZnO on FTO

$J_{sc} = 8.5 \text{ mA/cm}^2$

$V_{oc} = 0.216 \text{ V}$

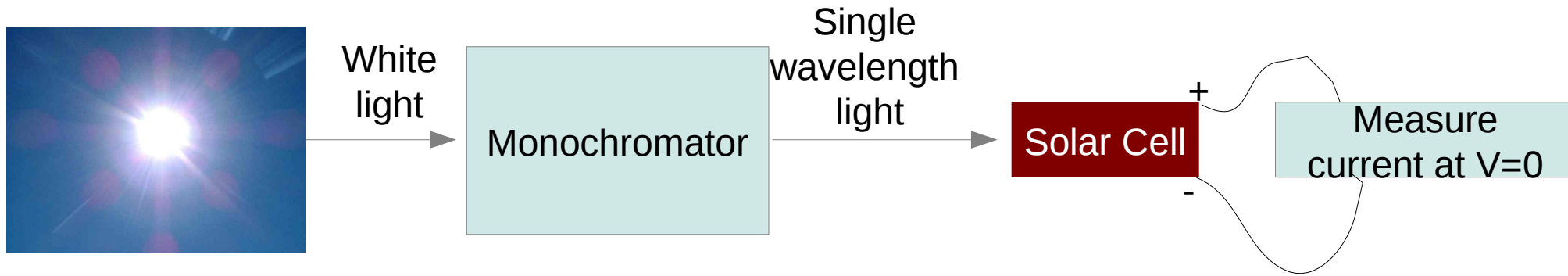
FF = 0.308

Eff = 0.57%

Nanowires increased carrier collection,  
indicated by increased  $J_{sc}$ .



# External quantum efficiency



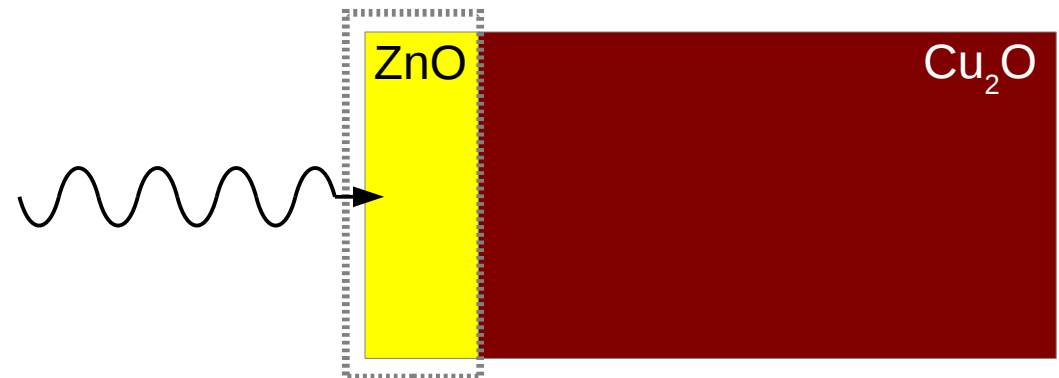
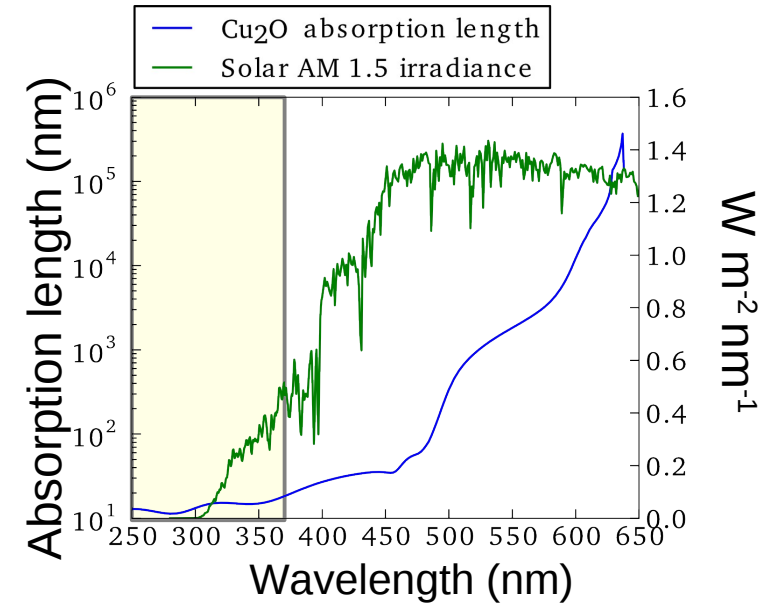
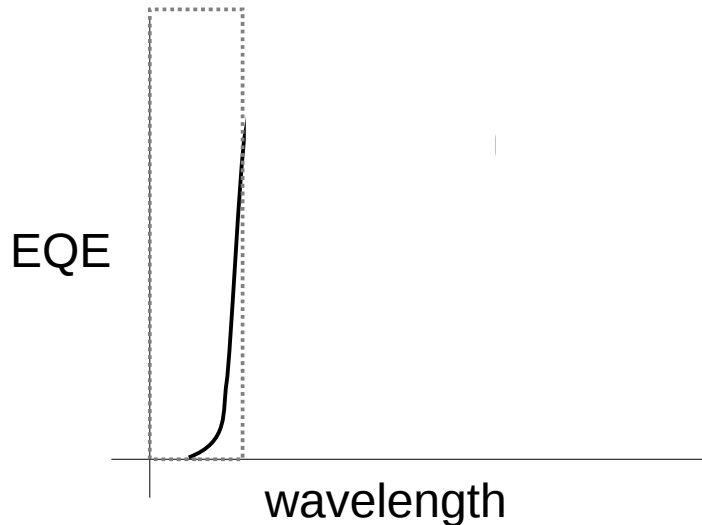
$$\text{EQE} = \frac{\# \text{ electrons collected } (\lambda)}{\# \text{ photons incident } (\lambda)}$$
$$= (\text{absorption probability}) \times (\text{carrier collection probability})$$



# External Quantum Efficiency

$$EQE = \frac{\# \text{ electrons collected } (\lambda)}{\# \text{ photons incident } (\lambda)}$$

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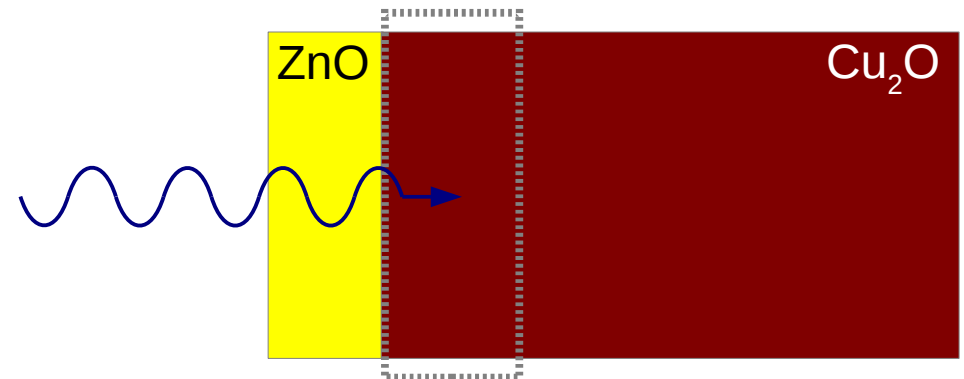
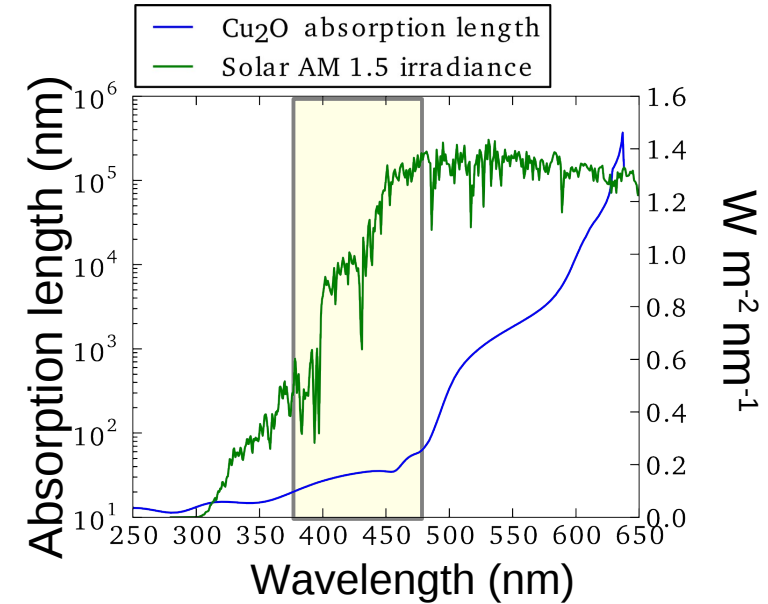
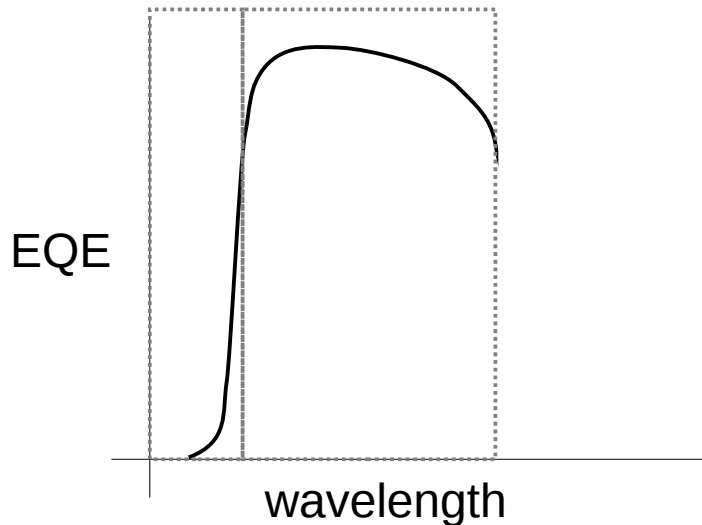




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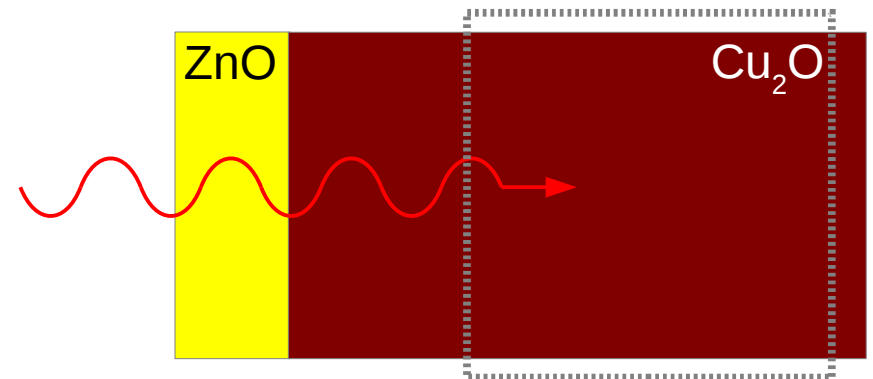
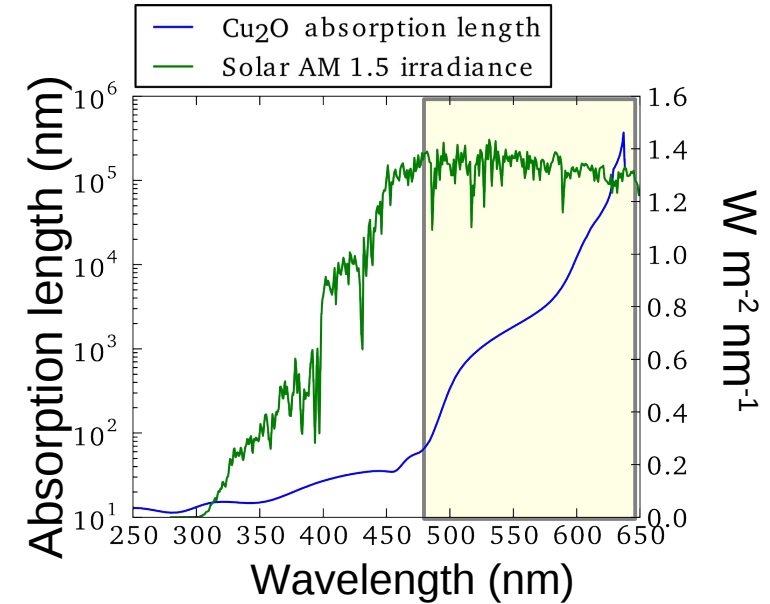
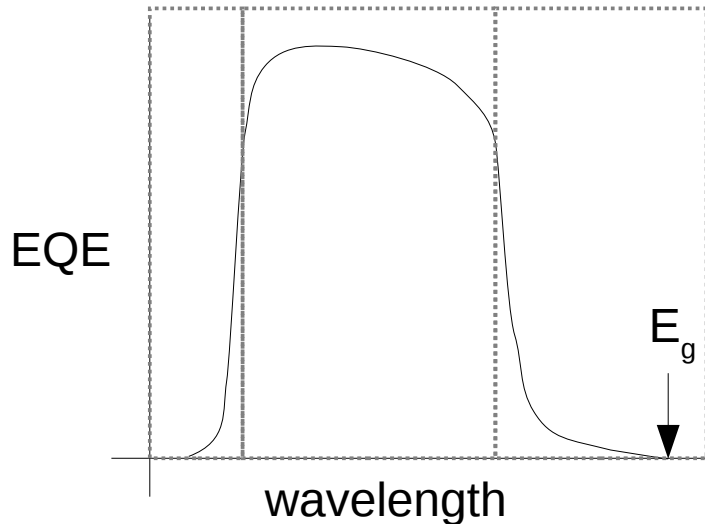




# External quantum efficiency

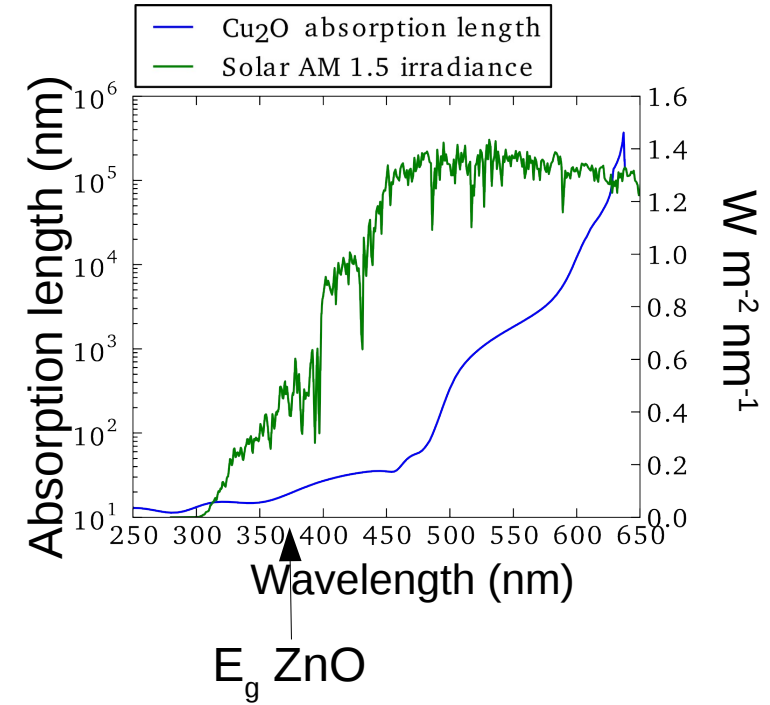
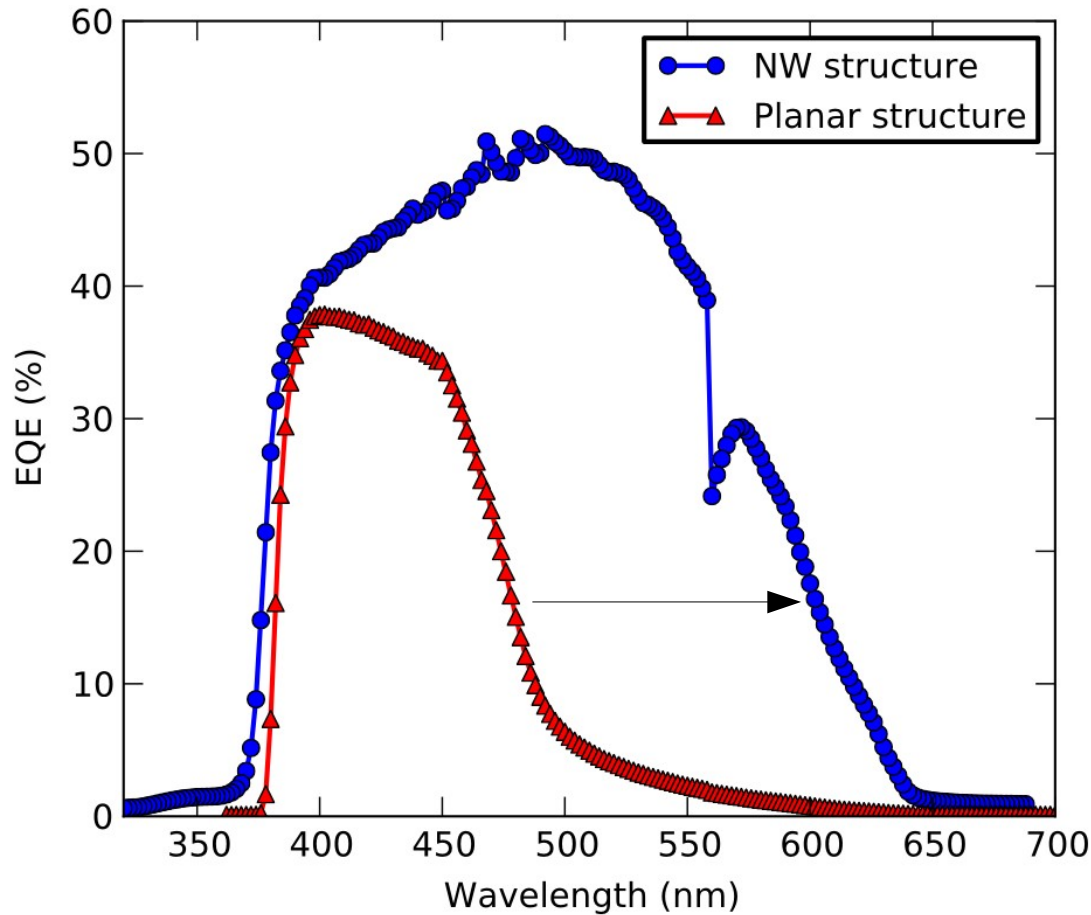
$$EQE = \frac{\# \text{ electrons collected } (\lambda)}{\# \text{ photons incident } (\lambda)}$$

$$= (\text{absorption probability}) \times (\text{carrier collection probability})$$





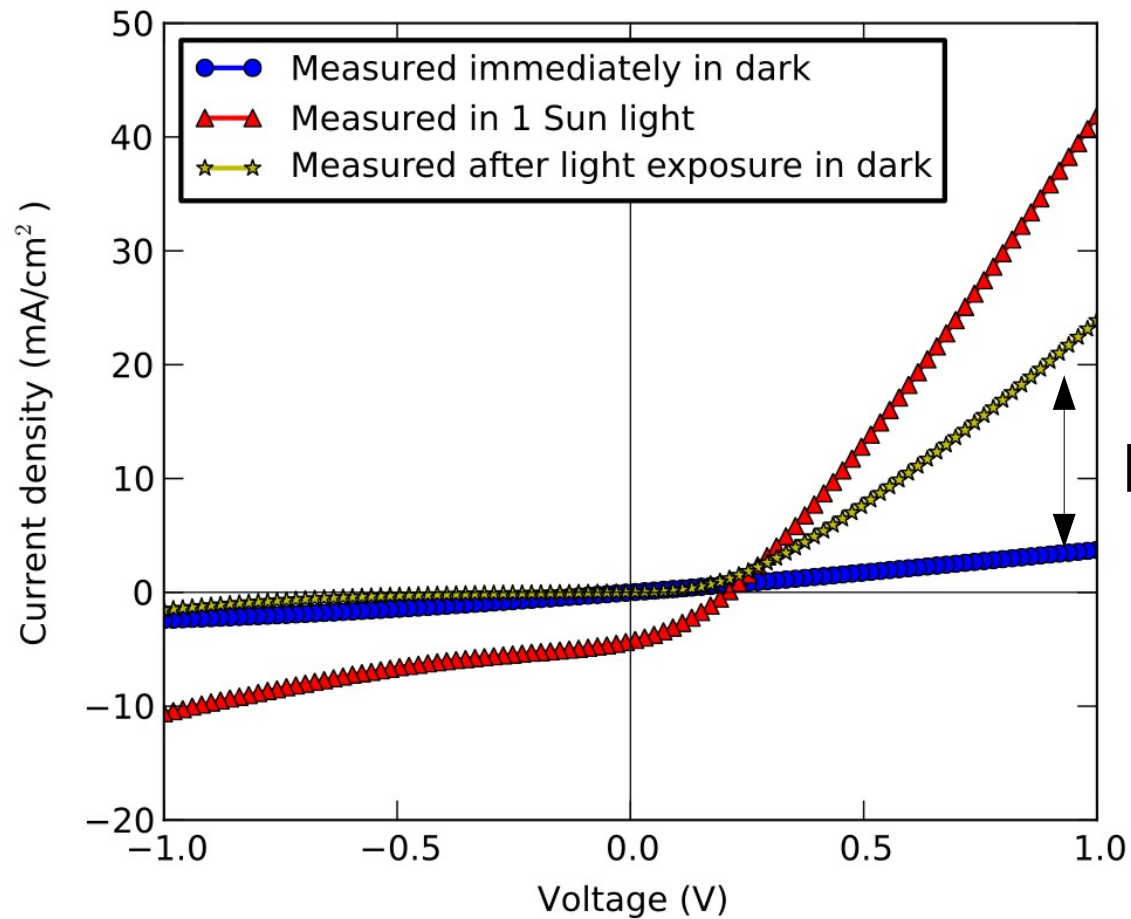
# Enhanced carrier collection: EQE



Nanowires increase carrier collection generated by long wavelength photons.



# Order of measurements is important



Not the same!



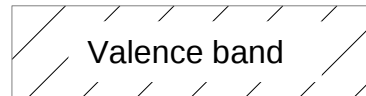
# Photoconductivity

$$\textit{conductivity} = \mu n e$$

$\mu$  : mobility

$n$  : electron concentration

$e$  : charge of electron





# Photoconductivity

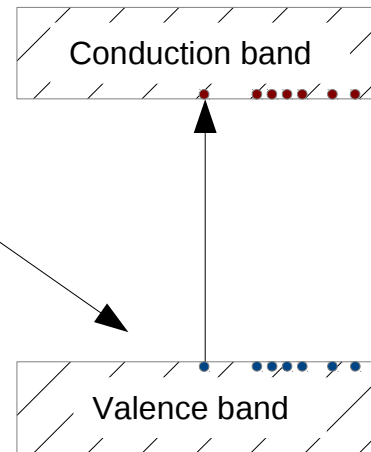
$$\text{conductivity} = \mu n e$$

$\mu$  : mobility

$n$  : electron concentration

$e$  : charge of electron

$$h\nu > E_g$$



$n$  increases.



# Photoconductivity

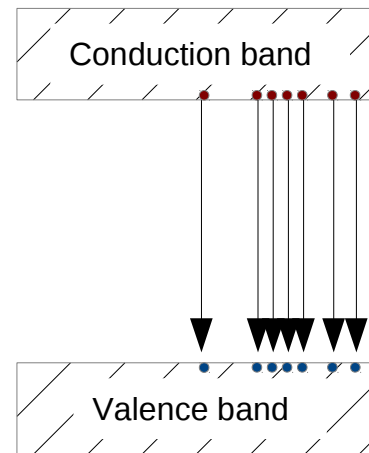
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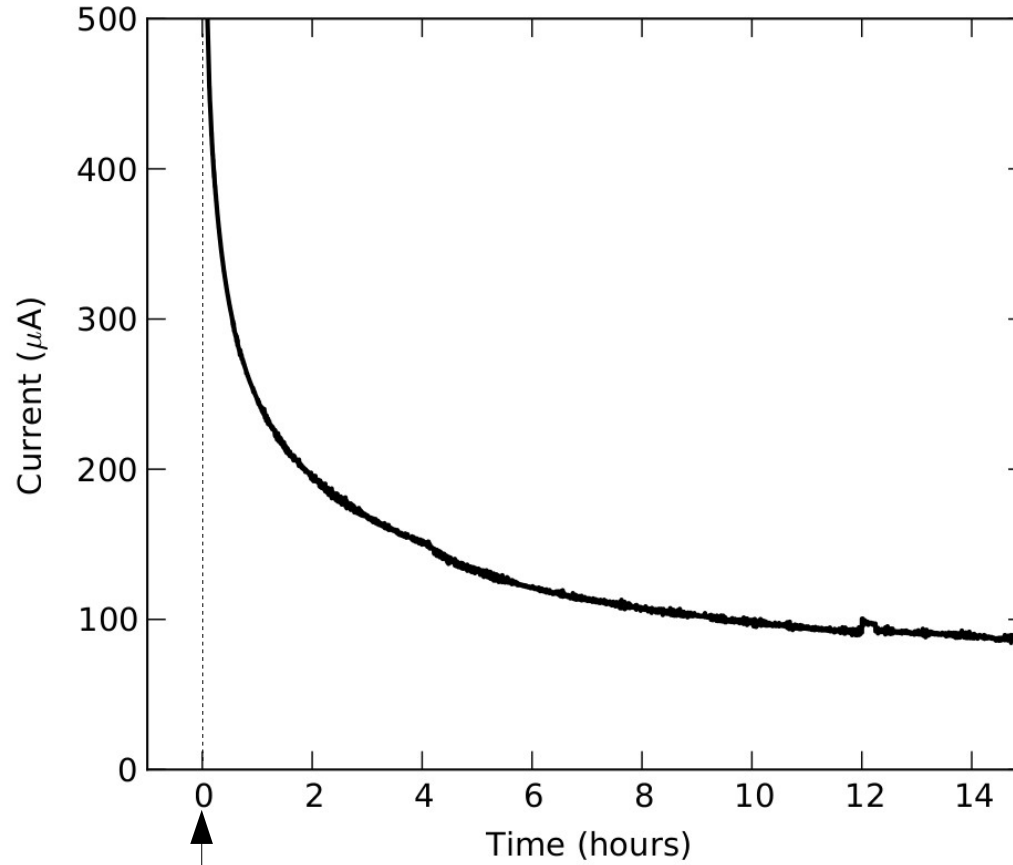
$n$  decreases.



Usually happens  $\sim 100$  ps for  $\text{Cu}_2\text{O}$ .<sup>1</sup>



# Decay in conductivity lasts hours



Turned off light at  $t=0$ .

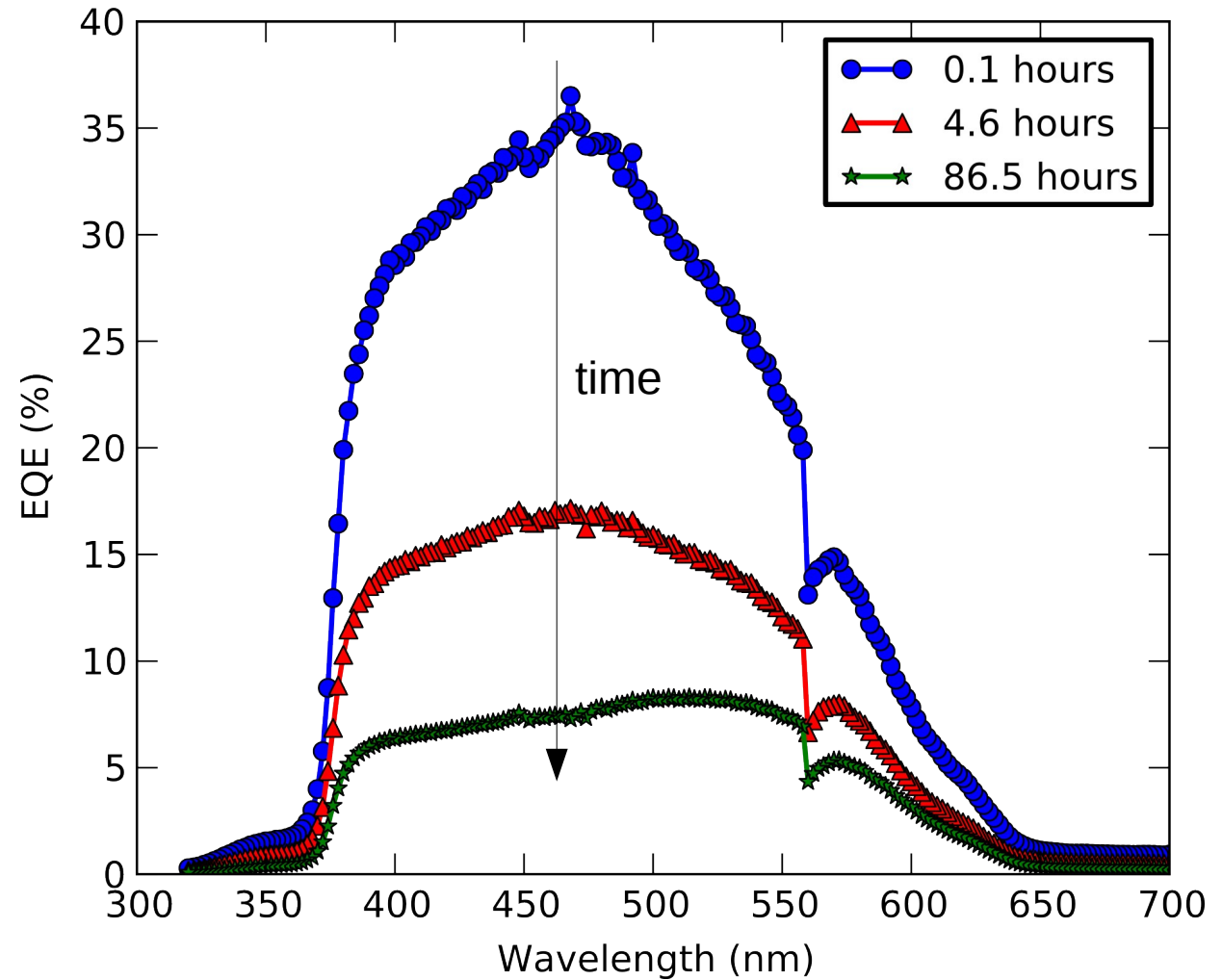


# EQE decays after initial exposure to light

Sample was illuminated with white light.

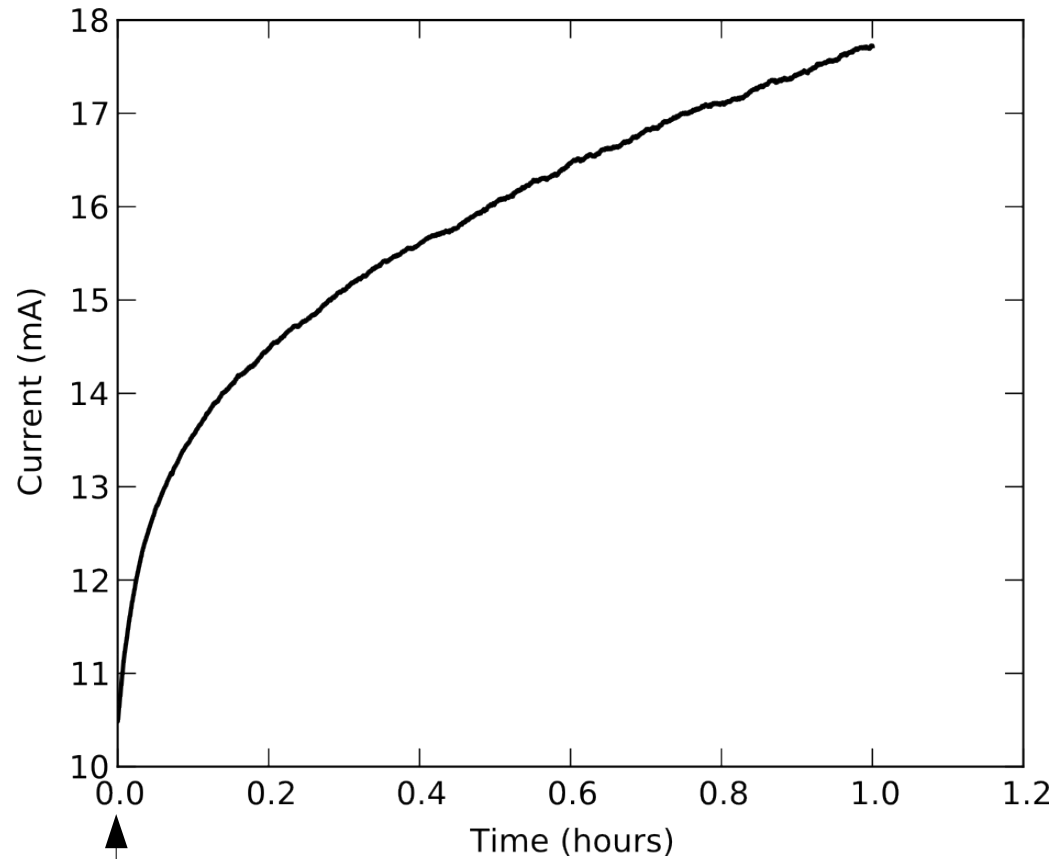
Illumination stopped at  $t=0$ .

EQE decayed after initial illumination.





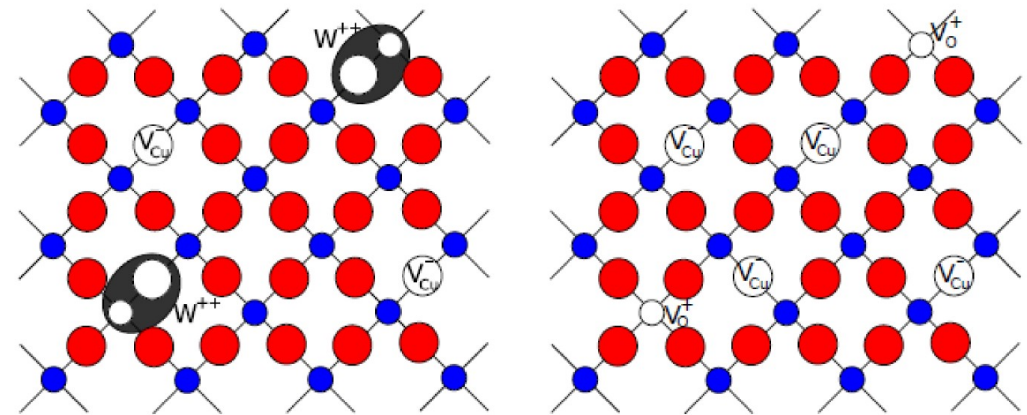
# Charging in conductivity is slow



Turned on light at  $t=0$ .

# Persistent Photoconductivity in $\text{Cu}_2\text{O}$

- Semiconductor remains conductive after removal of light – Persistent photoconductivity
- PPC mechanism in  $\text{Cu}_2\text{O}$  is due to dissociation and association of  $(V_{\text{Cu}}^- V_{\text{O}})$  complexes.



Before illumination

After illumination

● Oxygen

○ Oxygen vacancy

● Copper

○ Copper vacancy

In  $\text{Cu}_2\text{O}$ :

$$D(V_{\text{Cu}}) = 9.6 \times 10^{-10} \text{ cm}^2/\text{s}$$

In Si:

$$D(\text{B}) = 4.6 \times 10^{-54} \text{ cm}^2/\text{s}$$

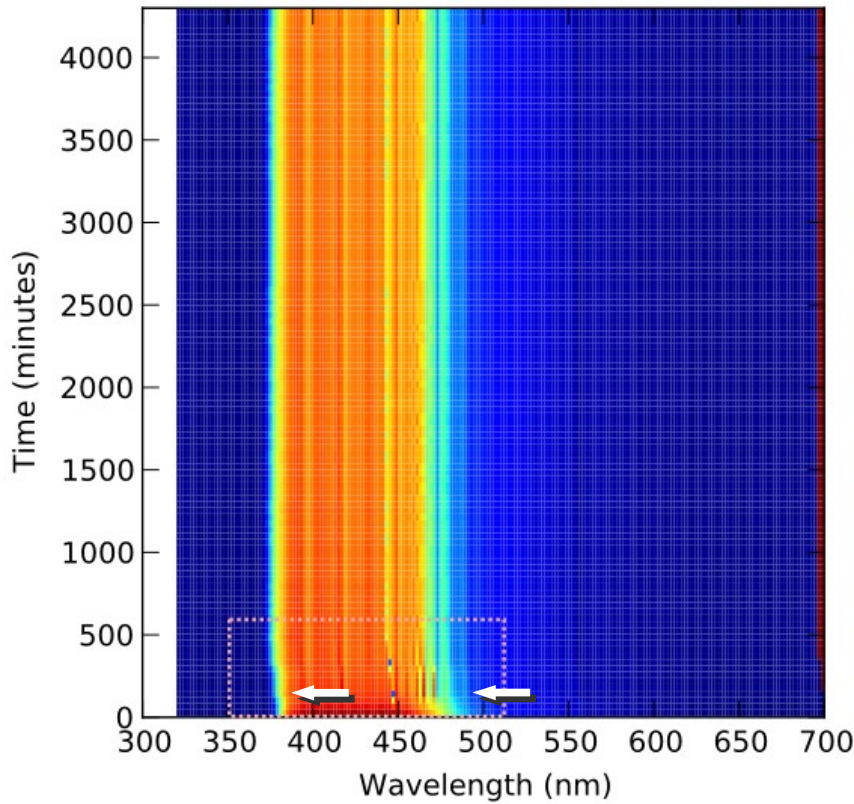
$$D(\text{P}) = 1.4 \times 10^{-49} \text{ cm}^2/\text{s}$$



# Shifting EQE

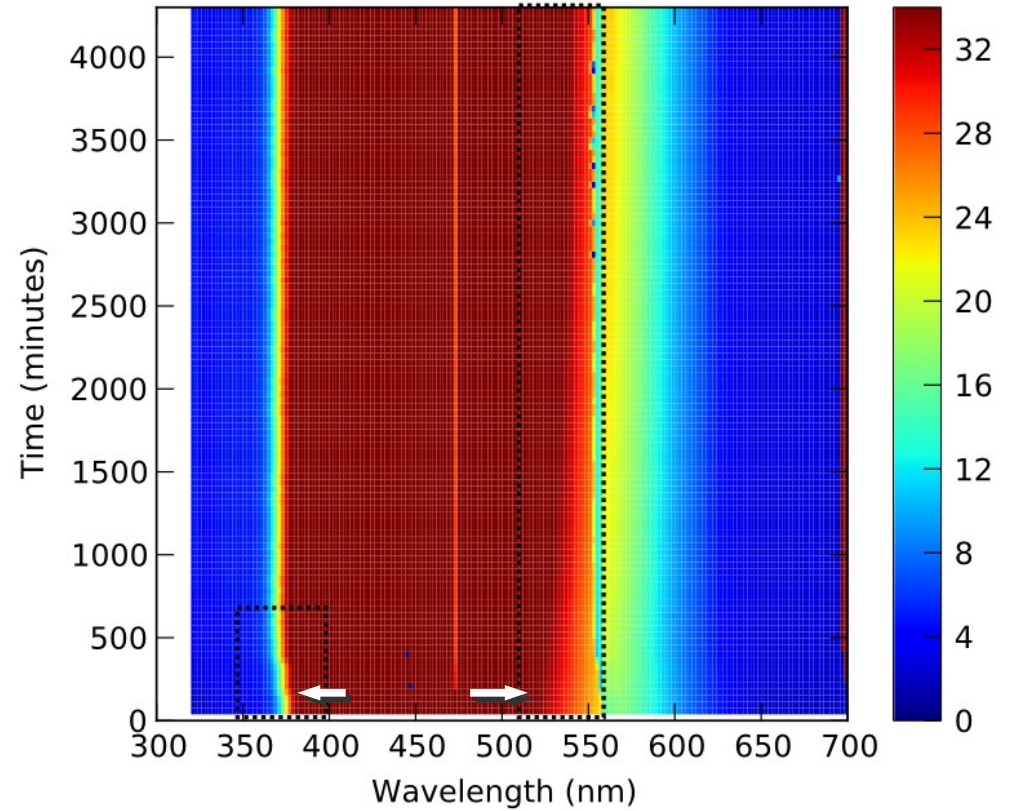
Planar

EQE(%)



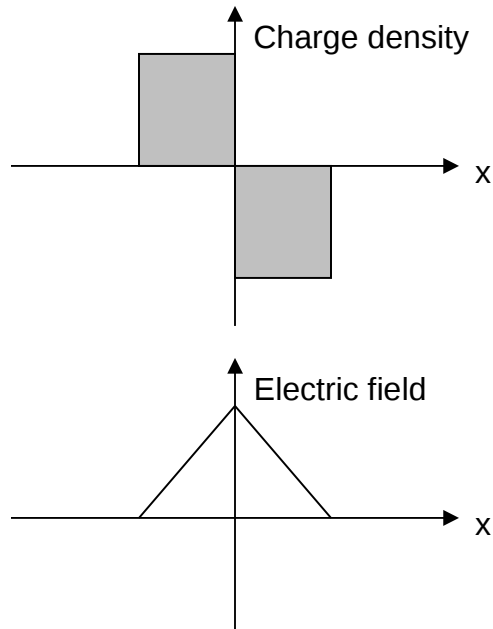
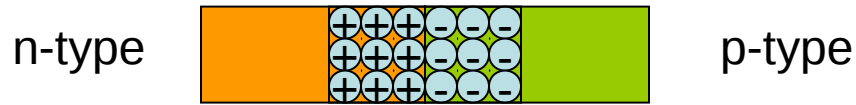
Nanowire

EQE(%)





# Mobile $V_{Cu}^-$ diffuses in $Cu_2O$



$$\frac{dE}{dx} = \frac{\rho}{\epsilon}$$

$V_{Cu}^-$  are highly mobile compared to other semiconductors.

Internal electric field drives  $V_{Cu}^-$  motion toward the interface.

In  $Cu_2O$ :

$$D(V_{Cu}^-) = 9.6 \times 10^{-10} \text{ cm}^2/\text{s}$$

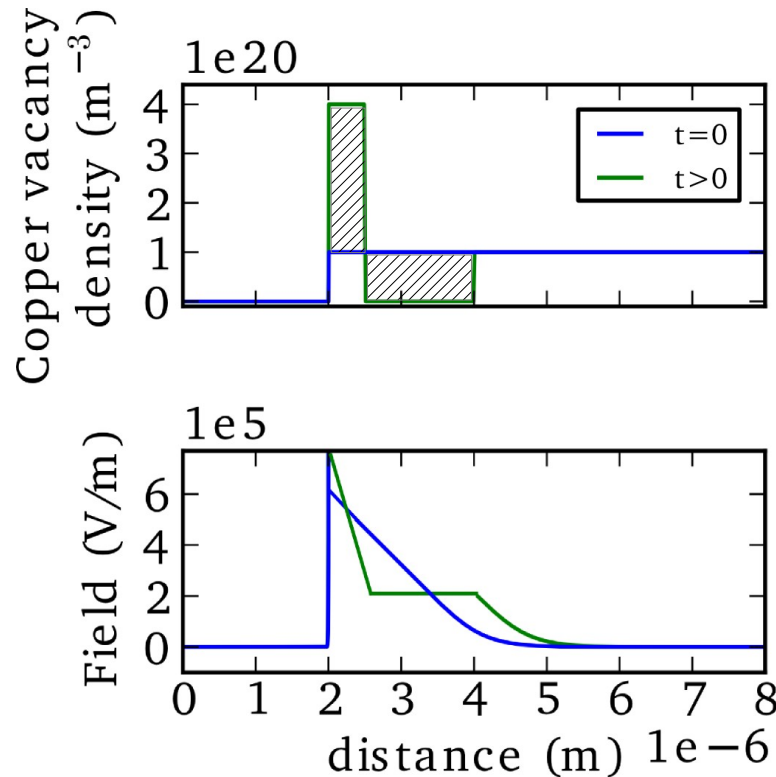
In Si:

$$D(B) = 4.6 \times 10^{-54} \text{ cm}^2/\text{s}$$

$$D(P) = 1.4 \times 10^{-49} \text{ cm}^2/\text{s}$$



# $V_{Cu}$ accumulates at the interface



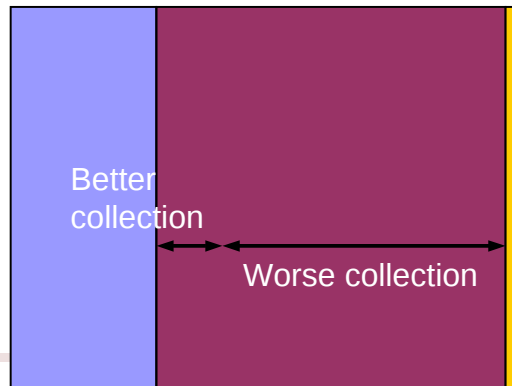
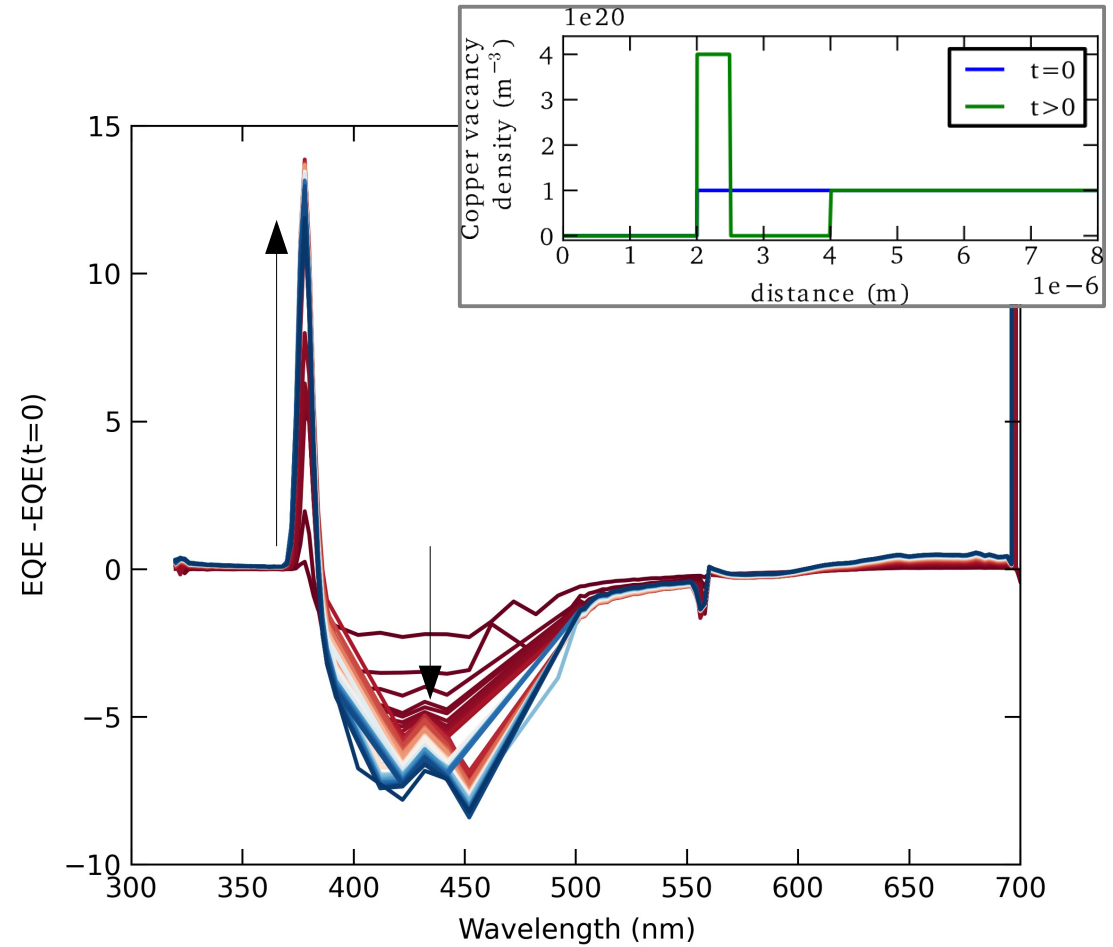
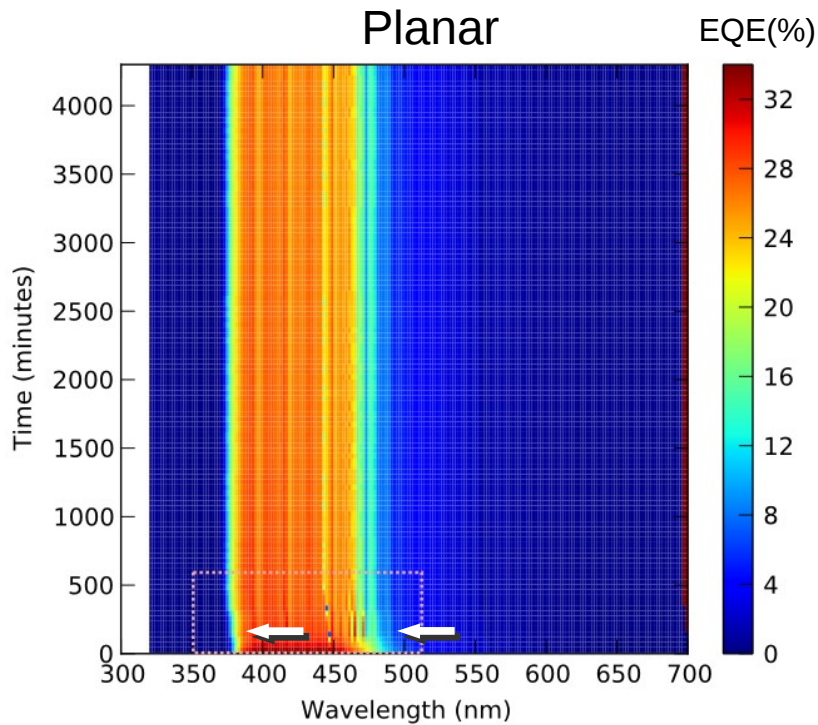
$V_{Cu}$  have moved to the interface, leaving behind a layer depleted of  $V_{Cu}$ .

Electric field is larger close the the interface.  
is smaller away from the interface.

Carrier collection is better close to the interface.  
worse farther from the inteface.

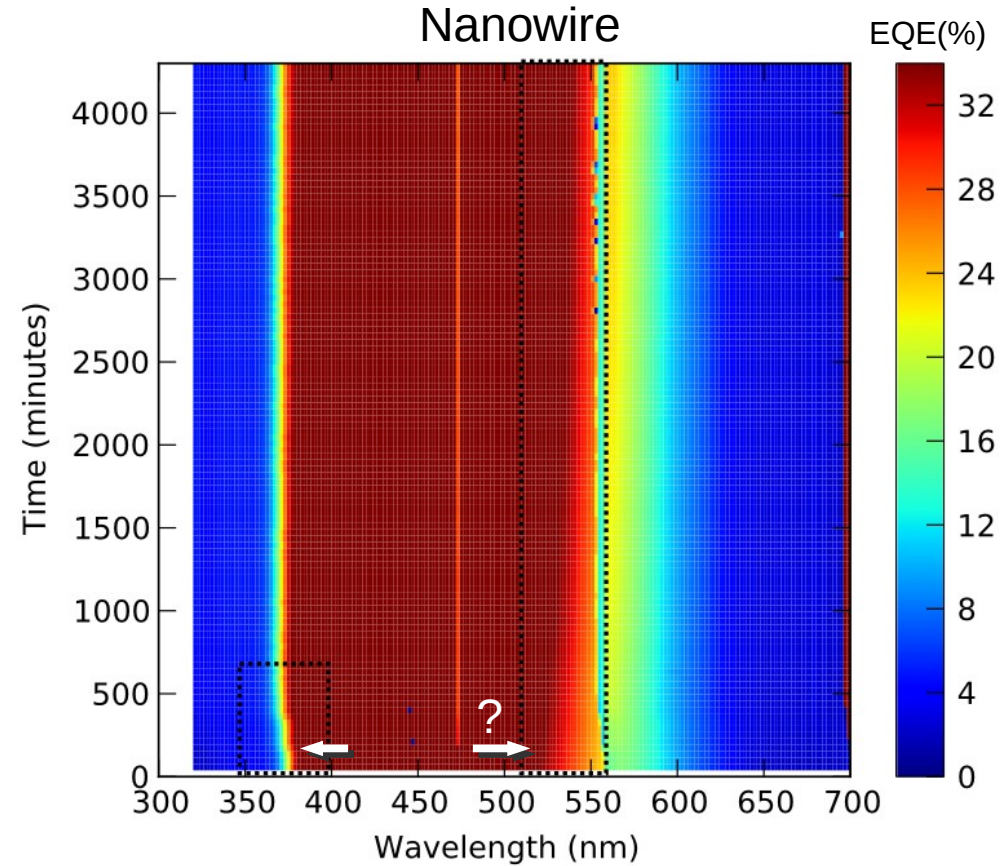
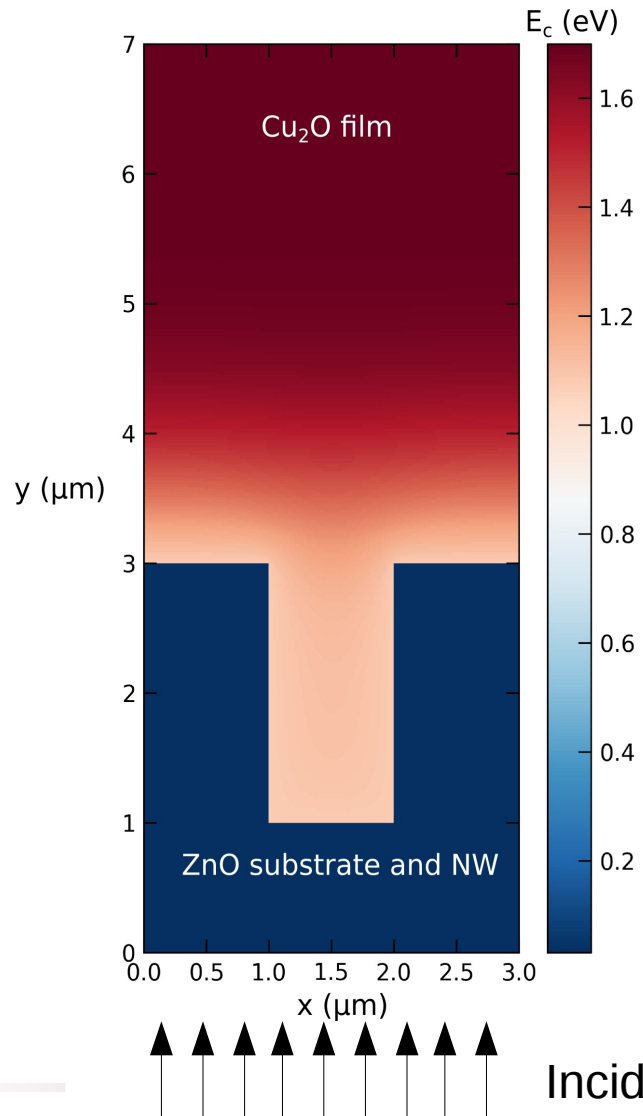


# Blue shift in EQE due to accumulation



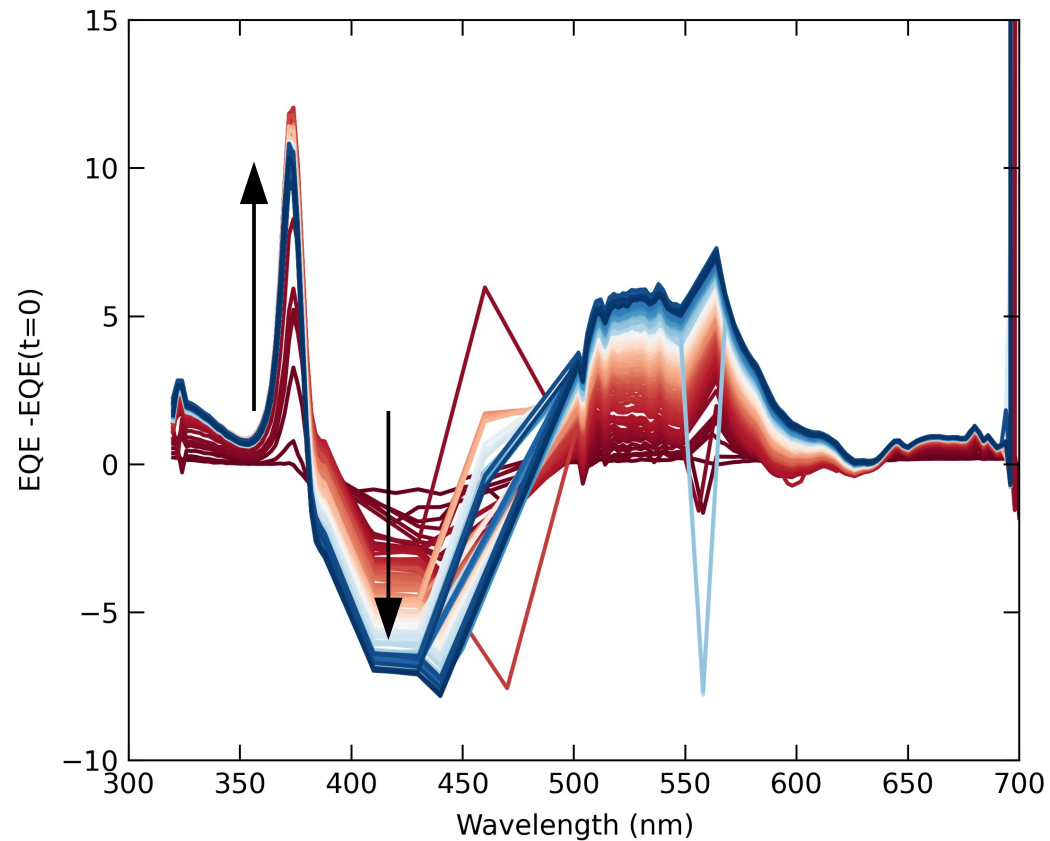
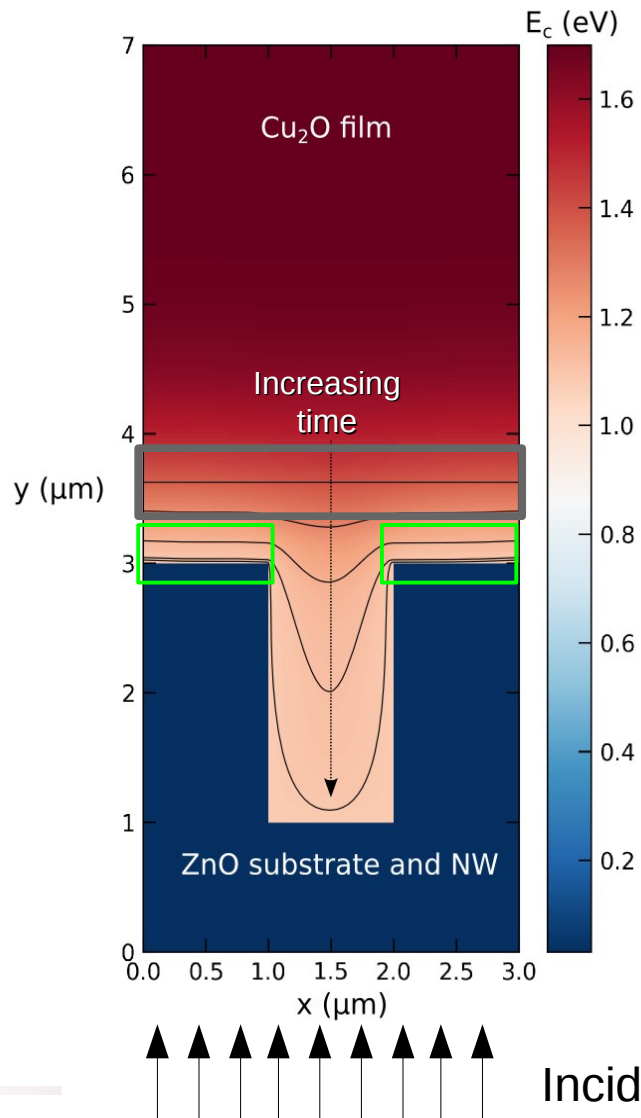


# Red shift in EQE for NWs due to accumulation





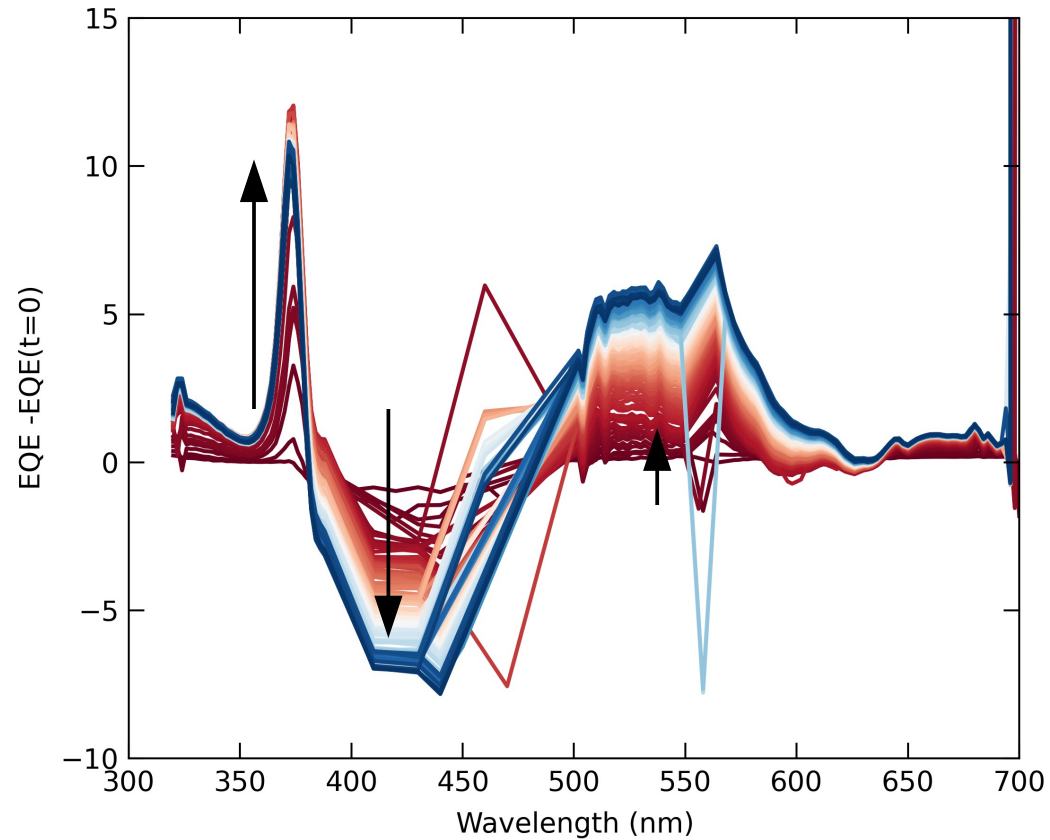
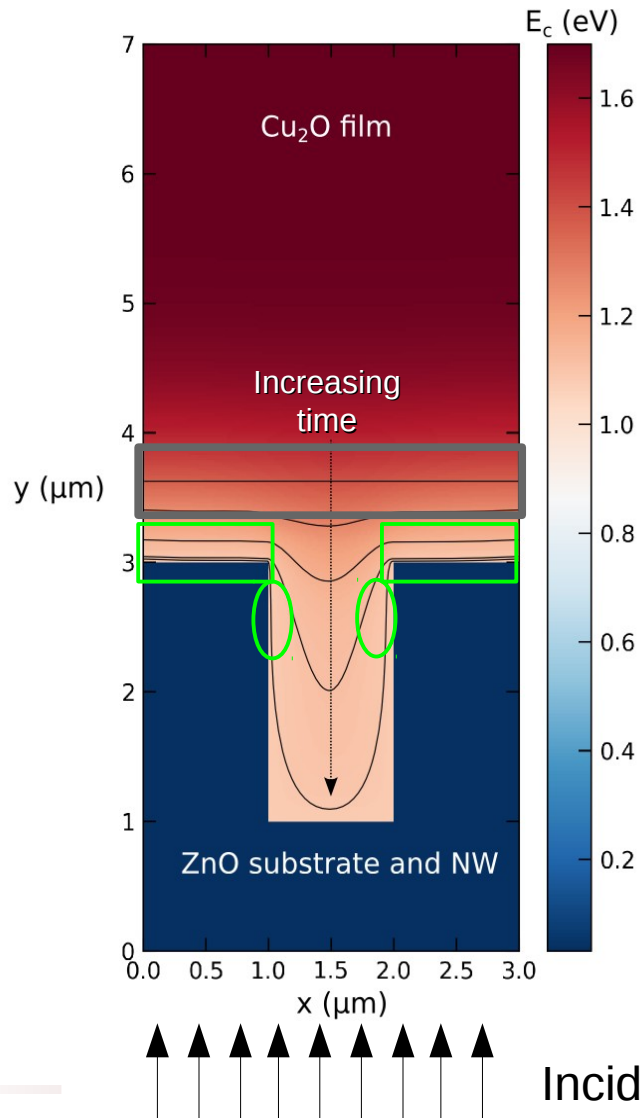
# Red shift in EQE for NWs due to accumulation



$V_{Cu}$  move into region between nanowires, creating higher electric field regions.



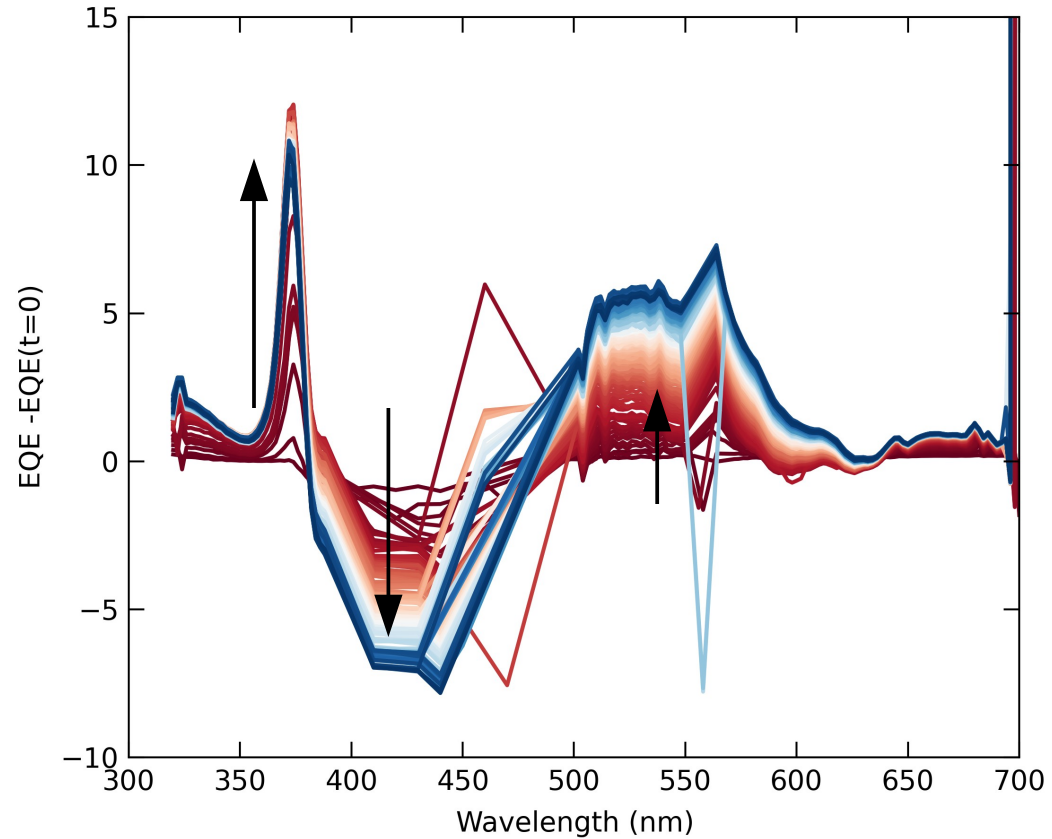
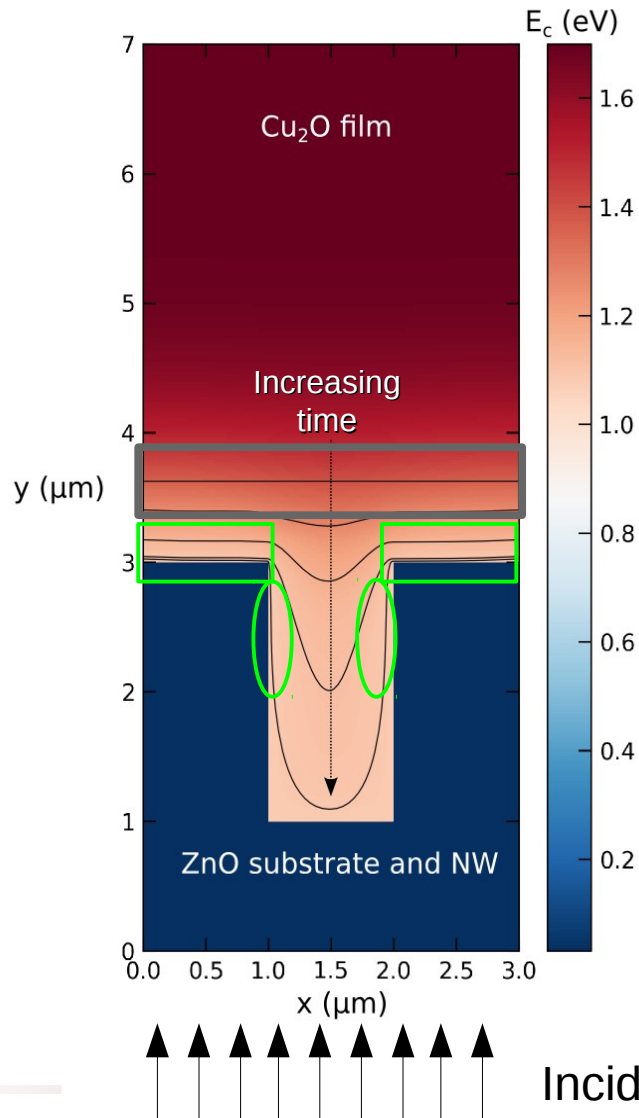
# Red shift in EQE for NWs due to accumulation



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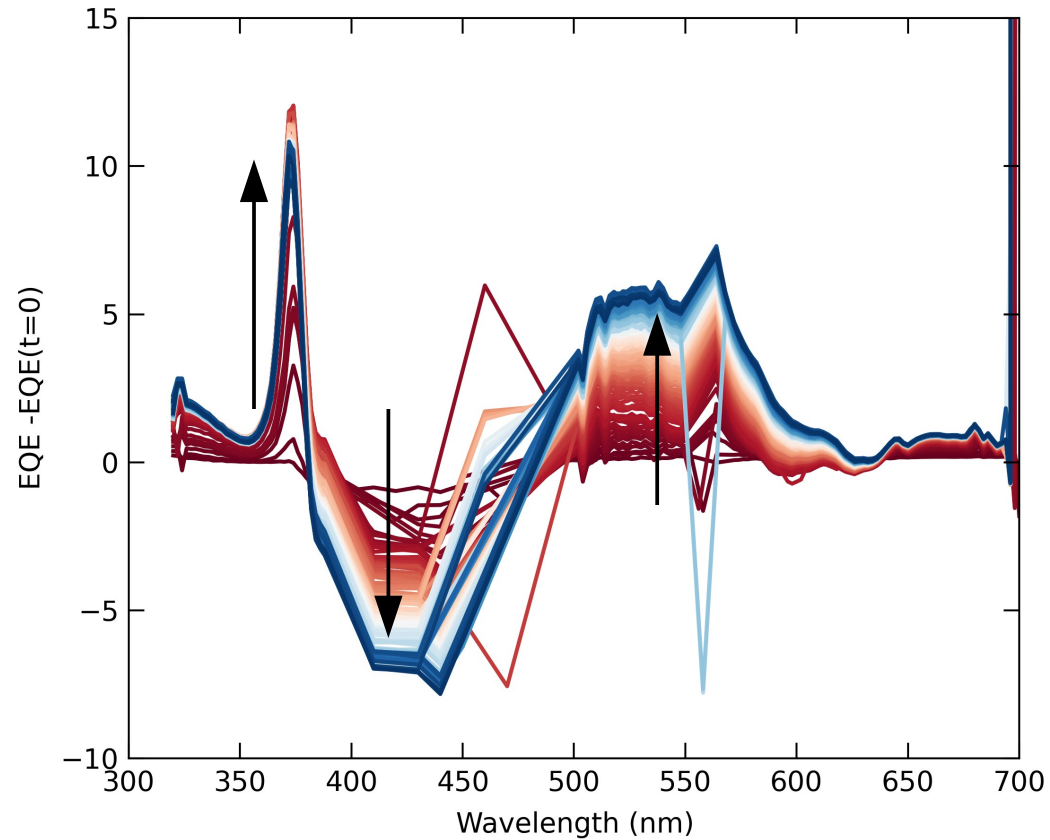
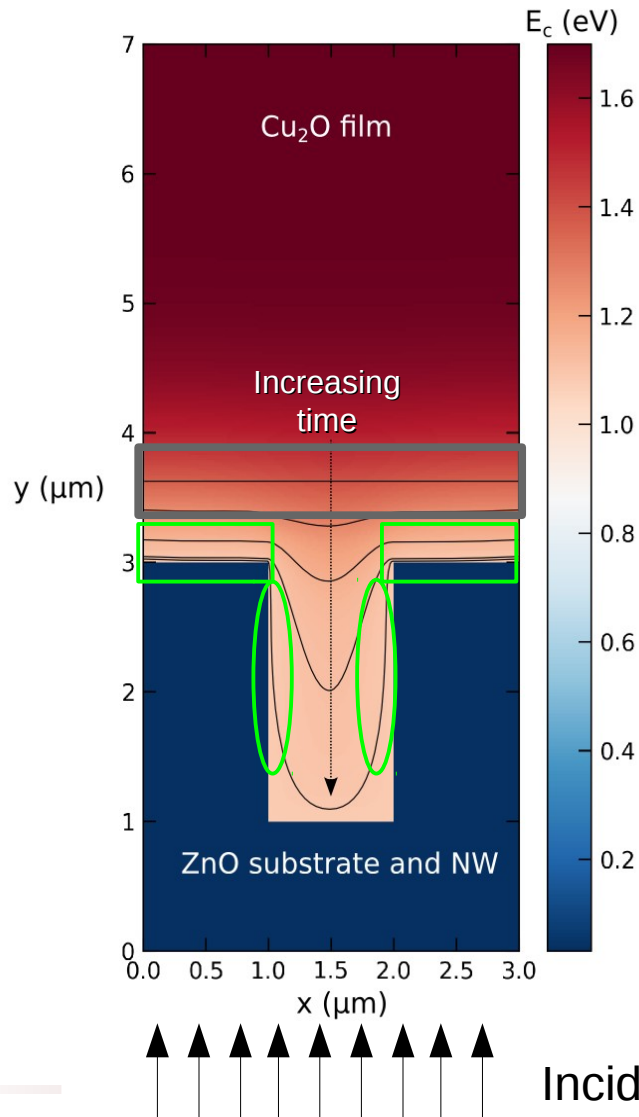
# Red shift in EQE for NWs due to accumulation



$V_{Cu}$  move into region between nanowires, creating higher electric field regions.



# Red shift in EQE for NWs due to accumulation

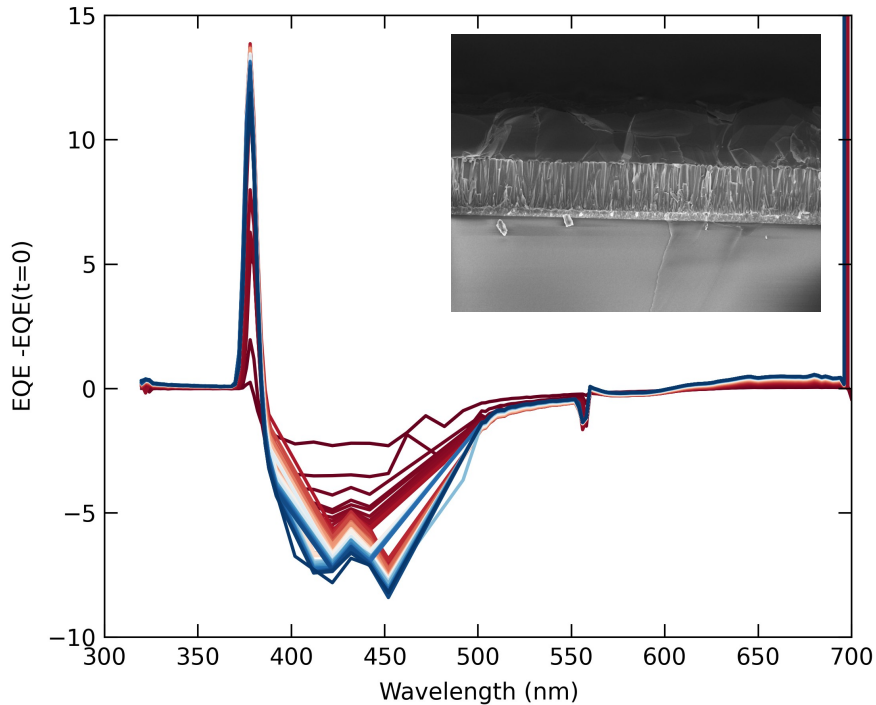


$V_{\text{Cu}}$  move into region between nanowires, creating higher electric field regions.

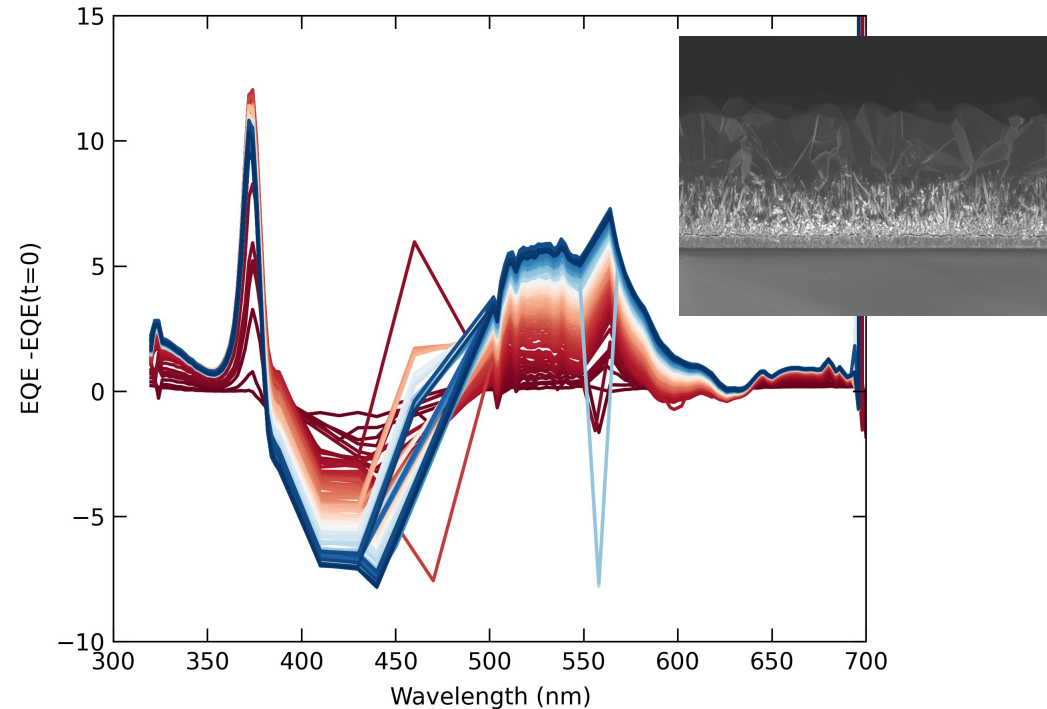


# Structure changes long term performance

Planar



Nanowire



- Long-term changes result from mobile dopant under the influence of the internal electric field.
- Structure of the solar cell can strongly influence the effect and the path of dopant migration.



# Summary and Implications

Controlled nanostructure with aqueous ZnO synthesis to obtain both nanowire and planar solar cells

Demonstrated improved carrier collection in the nanowire structure

Observed changes of EQE over time for both planar and nanowire devices, which were explained by moving dopants.



# Implications

Nanowire solar cells have potential in increasing carrier collection efficiency.

Long-term testing of solar cells is a critical part of the evaluation process.

Device design must account for possible dopant diffusion, especially in ionic semiconductors.

Creative dopant profiles may be engineered into the structure to enhance device performance.

Thank you for your attention!  
Questions?  
Acknowledgements to come....



# Acknowledgements

## Committee

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## Friends

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Josh Petrie

Shawn Mack

Natasha Vermaak

Yinan Zhang

## Apartment-mates

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Antrim

## Church communities

Santa Barbara Community Church

Highrock Church

## Family

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Chiou-Fu Wang

Christine Zgrabik

Alex Woolf

David Bracher

Tina Huang

Danqing Wang

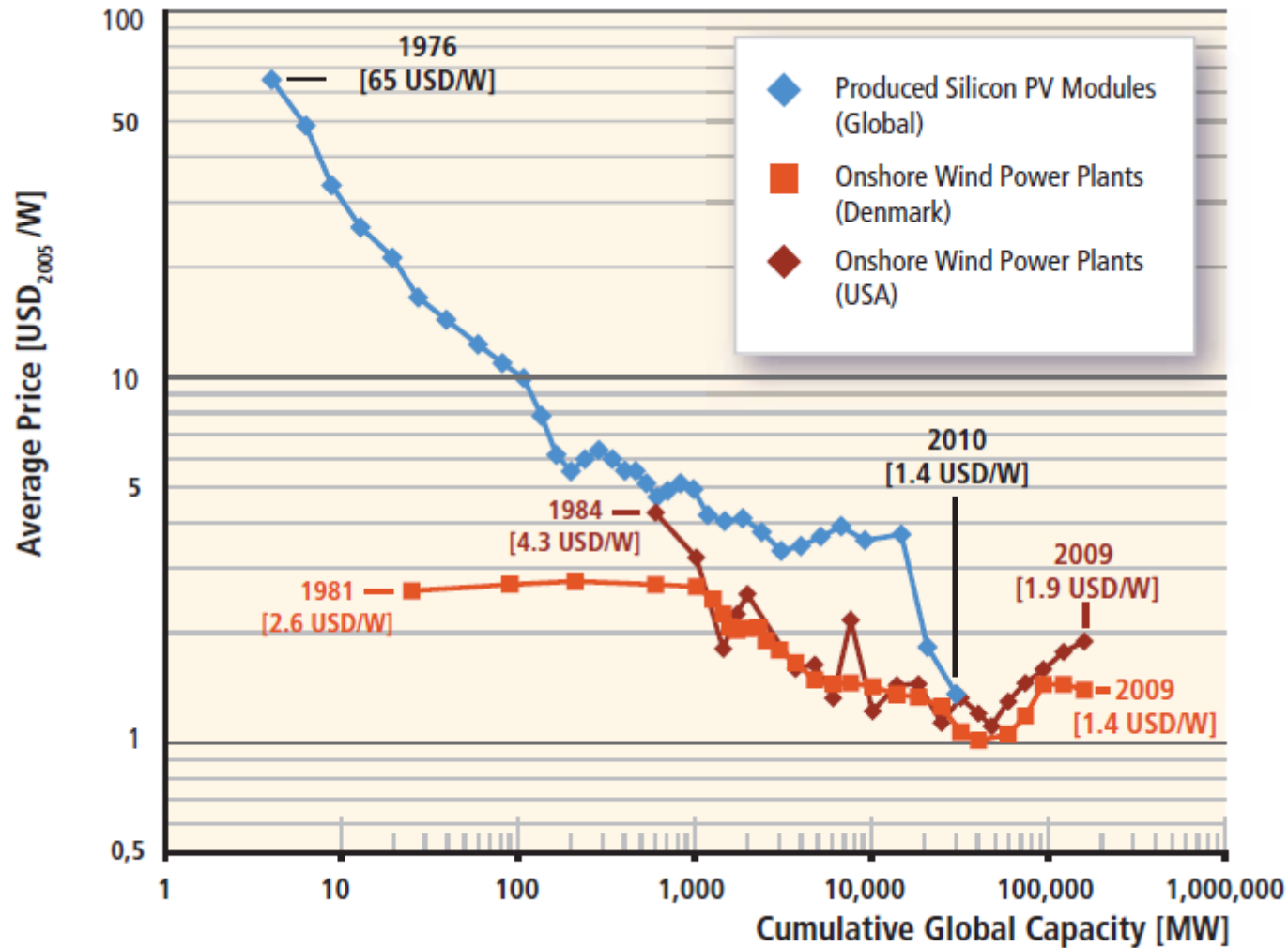
Alex Zhang

Andy Greenspon





# Price of wind increased!





$E_g = 2.0$  eV

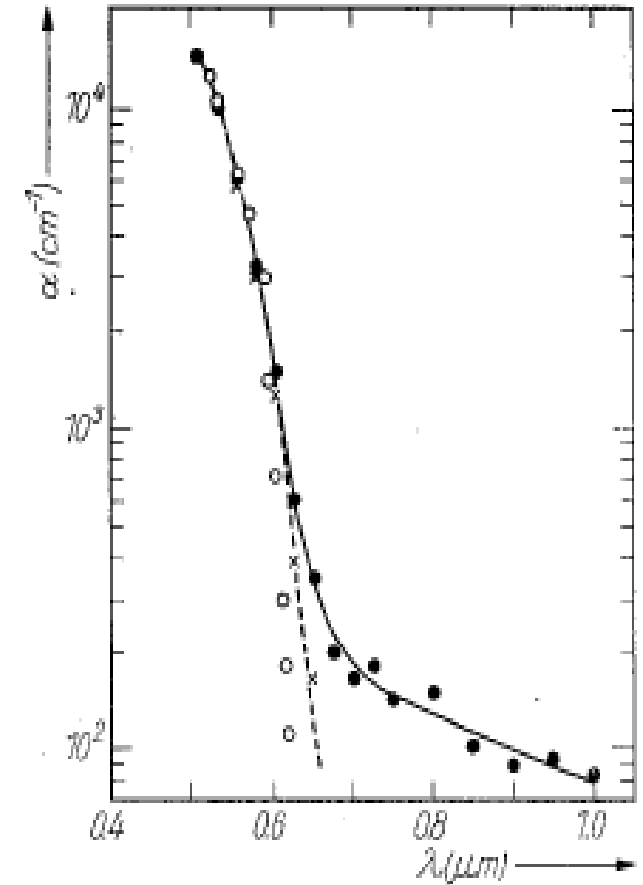
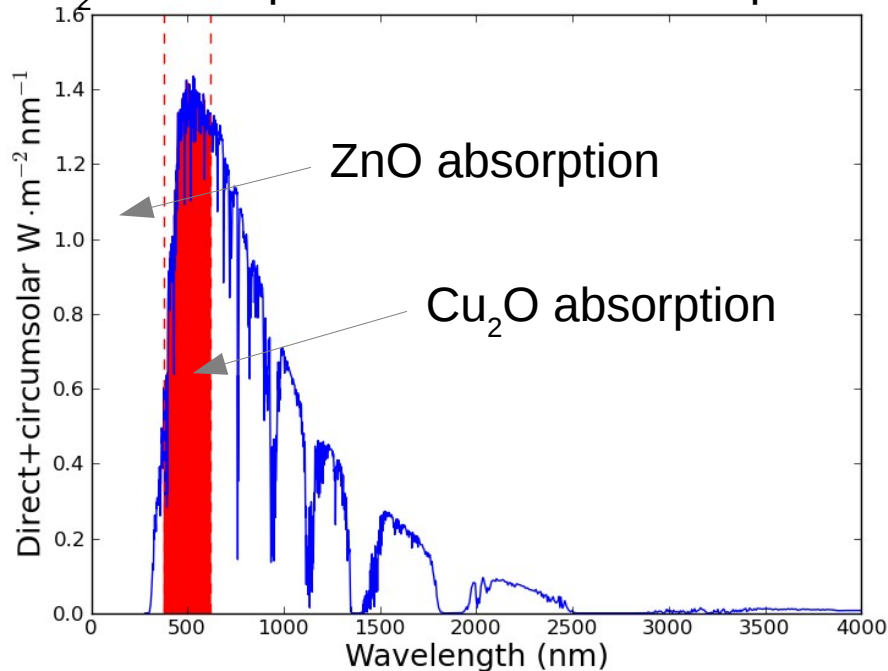
Direct band gap material

High absorption coefficients ( $\sim 10^4$  cm<sup>-1</sup>)

Typically p-type material, n-type very difficult

Electrodeposition possible for cheap production

Cu<sub>2</sub>O Absorption in AM 1.5 Solar spectrum



- Olsen, L. C., F. W. Addis & W. Miller (1982) Experimental and theoretical studies of Cu<sub>2</sub>O solar cells. *Solar Cells*, 7, 247-279.
- Rakshani, A. E. & J. Varghese (1987) Optical absorption coefficient and thickness measurement of electrodeposited films of Cu<sub>2</sub>O. *Physica Status Solidi a-Applied Research*, 101, 479-486.
- de Jongh, P. E., D. Vanmaekelbergh & J. J. Kelly (2000) Photoelectrochemistry of electrodeposited Cu<sub>2</sub>O. *Journal of the Electrochemical Society*, 147, 486-489.



# Solar cell circuit analysis

$$I = I_L - I_d - I_{sh}$$

$I$  = current across load

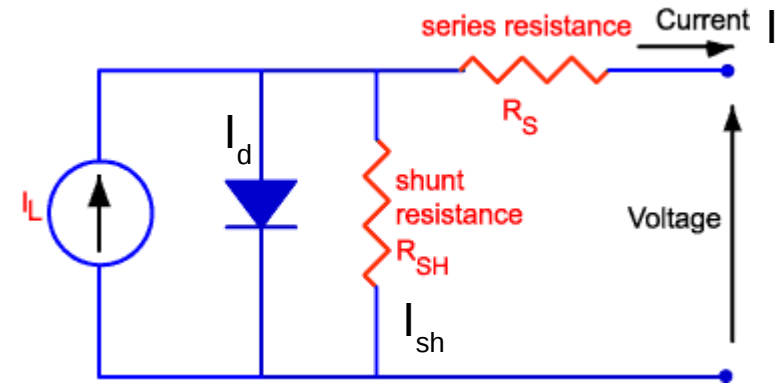
$I_d$  = diode current

$I_{sh}$  = shunt current

$$I_d = I_o \left( \exp\left(\frac{qV_j}{kT}\right) - 1 \right)$$

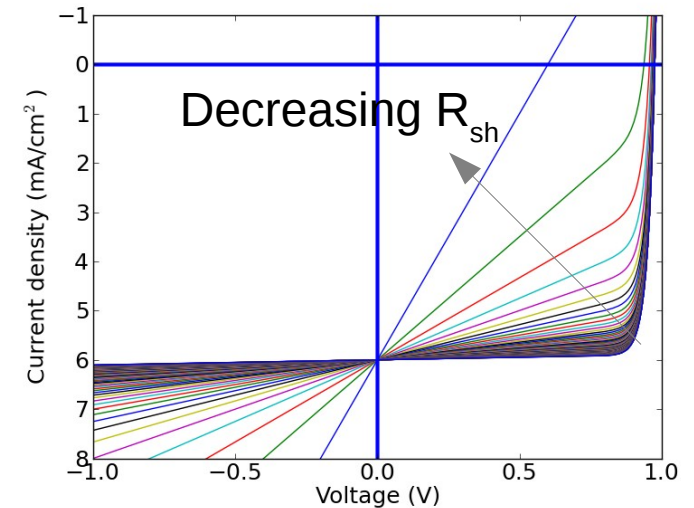
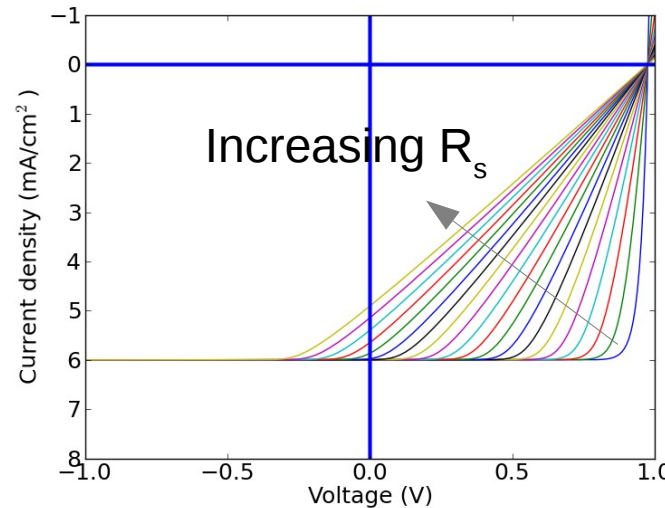
$$I_{sh} = \frac{V_j}{R_{sh}}$$

$$V_j = V + IR_s$$



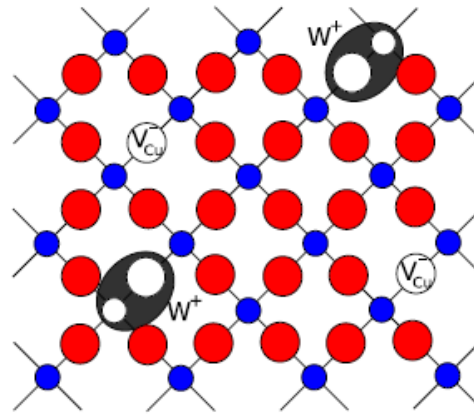
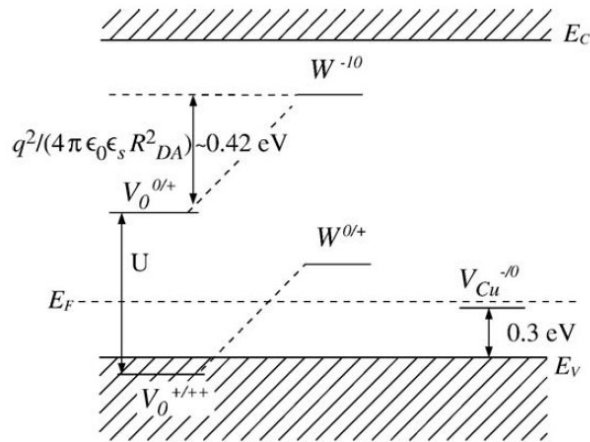
Shunt and series resistance modeled in solar cell circuit.<sup>1</sup>

## Effect of $R_s$ and $R_{sh}$ on Solar Cell JV curves

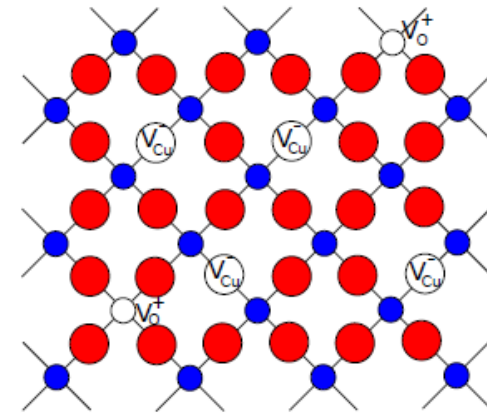




# PPC (complexing model)



Before illumination

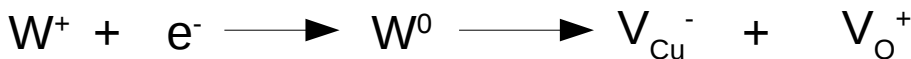


After illumination

Before illumination:



During illumination:



Illumination is stopped:

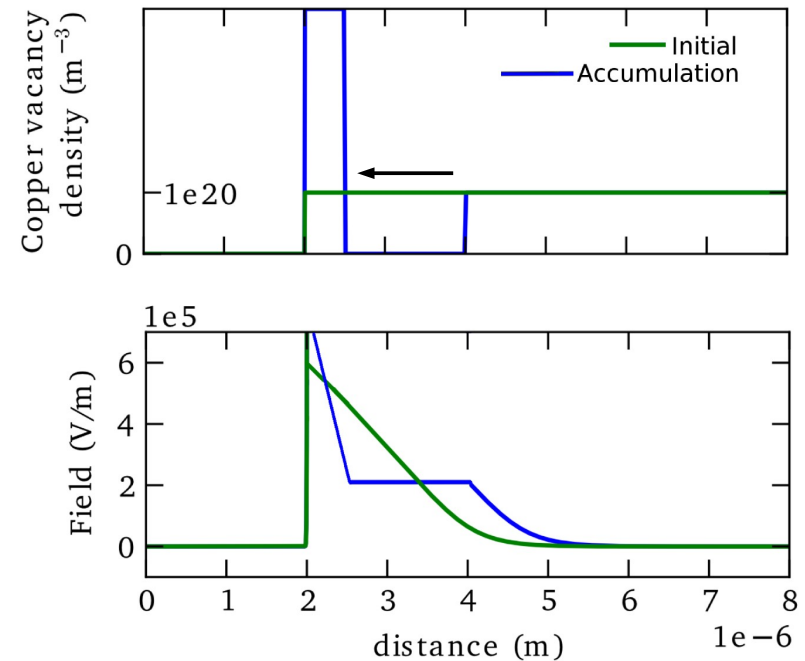
$V_O^+$  cannot be restored to  $V_O^{++}$  through hole capture, because  $V_O^+$  energy level is above Fermi level.

$V_{Cu}^-$  must diffuse back to  $V_O^+$  to make  $W^0$ , which can capture a hole to become  $W^+$ .



# Passivation of $V_{Cu}^-$ at interface

$H^+$  from ZnO  
complexes with  $V_{Cu}^-$   
 $V_{Cu}^-$  oxidizing to  $V_{Cu}^{2-}$   
and forming CuO at  
interface





# $\text{Cu}_2\text{O}$ darkens after long exposures to light

